# **ES-07**

# EFFECTS OF Mg DOPING AND SINTERING TEMPERATURE ON THE MAGNETORESISTANCE OF SINTERED Fe<sub>3</sub>O<sub>4</sub> FERRITES

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### Introduction

Currently, the magnetoresistance (MR) of pure Fe<sub>3</sub>O<sub>4</sub> has been investigated by few groups 11-21. The MR value of Fe<sub>2</sub>O<sub>1</sub> in these studies was found very small at room temperature. (MR =  $0 \sim 1.7$  %). For the sintered Fe<sub>3</sub>O<sub>4</sub> ferrite, its MR value was dependent on the microstructure and sintering conditions. In this work, we improve MR value of the sintered Fe<sub>3</sub>O<sub>4</sub> ferrite at room temperature to about 7.3 % with small amount of Mg addition. The effects of Mg content, sintering temperature (Ts), and microstructure on its magnetoresistance at room temperature were investigated.

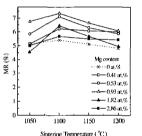
Experimental procedure The high-purity Fe<sub>3</sub>O<sub>4</sub> powder was mixed with various amount of MgO powder (0~25mol.%). After fully mixed, the mixed powder was compressed into a pellet (10 mm diameter, 1 mm thick) under a pressure of 53393 lb/in<sup>2</sup> then sintering in argon atmosphere at temperatures between 1050 °C and 1200 °C for 3 hours. Composition of the sintered samples was analyzed by energy disperse spectrometer (EDS). Crystalline structure of the sintered was analyzed by energy disperse spectrometer (EDs), constanting and the of the sintered sample was determined by using X-ray diffractometer (XRD) with CuKa radiation. Microstructure was observed with a scanning electron microscope (SEM). The Fe<sup>+</sup> and Fe<sup>+</sup> ion contents of the sintered sample were examined by chemical titration method. The magnetoresistance of the sintered sample was measured at room temperature with the four-probe method and the maximum applied field was 9 kOe.

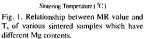
## Results and discussion

It is found that the Fe<sub>3</sub>O<sub>4</sub> and MgFe<sub>2</sub>O<sub>4</sub> phases are co-existed in all samples which sintered at temperatures between 1050  $^{\circ}$ C and 1200  $^{\circ}$ C from their X-ray diffraction patterns. This indicates that the MgO oxide reacts with some Fe<sub>3</sub>O<sub>4</sub> to form MgFe<sub>2</sub>O<sub>4</sub> ferrite during sintering. From the analysis of SEM element mapping, we find that the Mg ions are dispersed uniformly in all sintered samples. Figure 1 shows the relationship between MR value and T, of various sintered samples with different Mg contents at room temperature. The optimum  $T_s$ , which means that maximum MR value can be obtained is about 1100  $\sqcup$  for all samples, as shown in Fig. 1. Figure 2 shows the relationship between Mg content and MR value at room temperature of the sintered sample which was sintered at 1100  $\Box$ . The maximum MR value is about 7.3 % which occurs at Mg content of 0.93 at.%. The grain size of this sample is about 2.5  $\mu$ m and its density is 4.95 g/cm<sup>3</sup> which is about 94 % theoretical density of the pure Fe<sub>3</sub>O<sub>4</sub> (That is 5.24 g/cm<sup>3</sup>).

Chemical titration analysis of  $Fe^{2+}$  and  $Fe^{3+}$  ions contents indicates that the

nonstoichiometric phases of Fe<sub>3</sub>O<sub>4+X</sub> and Mg<sub>2</sub>Fe<sub>2</sub>O<sub>4-X</sub> are also existed in the sintered sample. The electric resistivity poly of the sintered sample was measure by the four-probe method at temperatures between 80 K and room temperature. The relationship between log p and  $T^{-1/2}$  of sintered samples with different Mg contents was shown in Fig.3.  $T^{1/2}$  and log p are linearly related for all samples, implying that the magnetoresistance effect of these sintered samples is spin-dependent tunneling [3]. This means that the insulators Fe<sub>2</sub>O<sub>4</sub>, Mg<sub>2</sub>Fe<sub>2</sub>O<sub>4</sub>, and Mg<sub>2</sub>Fe<sub>2</sub>O<sub>4</sub>, provide the tunneling barriers for enhancing the MR value. The length of spin-dependent tunneling barrier becomes too large and the MR value decreases with increasing Mg content as the Mg doping amount is higher than 0.93 at.%, as shown in Fig.2.





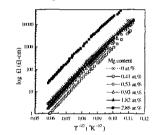
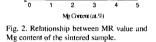


Fig. 3. log  $\rho$  vs. T<sup>-1/2</sup> of various samples which were sintered at 1100 °C.



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References

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