



## Characterization of cobalt oxides studied by FT-IR, Raman, TPR and TG-MS

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### ABSTRACT

The as-prepared cobalt oxide (assigned as  $\text{CoO}_x$ ) was fabricated by precipitation–oxidation from aqueous cobalt nitrate solution using sodium hydroxide and oxidation with hydrogen peroxide. Another series of pure cobalt oxides was refined by the decomposition of  $\text{CoO}_x$  in a nitrogen environment at temperatures of 280, 450 and 950 °C (D-280, D-450 and D-950, respectively). Phase transformation, structural properties and red-ox properties were characterized by thermogravimetry-mass spectrometry (TG-MS), X-ray diffraction (XRD), infrared spectroscopy (IR), Raman spectroscopy and temperature-programmed decomposition/reduction (TPD/TPR). Analysis of the thermal behavior on  $\text{CoO}_x$  revealed that a series of pure cobalt oxide with particle sizes of 10–20 nm could be obtained easily. The results demonstrated that the refined samples D-280, D-450 and D-950 were  $\text{CoO}(\text{OH})$ ,  $\text{Co}_3\text{O}_4$  and  $\text{CoO}$ , respectively.

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## 1. Introduction

Cobalt oxide has a wide range of applications in various industrial sectors, including in rechargeable batteries [1], as a catalyst of the abatement of CO [2], as a magnetic material [3] and in CO sensors [4]. Five species of cobalt oxide [ $\text{CoO}_2$ ,  $\text{Co}_2\text{O}_3$ ,  $\text{CoO}(\text{OH})$ ,  $\text{Co}_3\text{O}_4$  and  $\text{CoO}$ ] have been reported [4–9]. However, cobalt oxide with a valence of more than three is unstable in the natural environment. Other cobalt oxides [ $\text{Co}_3\text{O}_4$  and  $\text{CoO}$ ] are more stable and useful in industry. Cobalt oxyhydroxide,  $\text{CoO}(\text{OH})$ , has a hexagonal structure in which a divalent metal cation is located at an octahedral site which is coordinated by six hydroxyl oxygen. Well-spread  $\text{CoO}(\text{OH})$  can be used as the conductive network in rechargeable alkaline batteries [5].  $\text{CoO}$  is an antiferromagnetic material whose magnetic characteristics [10,11] and application of gas-sensors [12,13] have been extensively studied.  $\text{CoO}$  nanocomposite film in gas-sensing.  $\text{Co}_3\text{O}_4$  exhibited high catalytic activity in CO oxidation [14–16]. The spinel trivalent cobalt tetraoxide,  $\text{Co}_3\text{O}_4$ , has an energy band-gap of 1.4–1.8 eV [17] that can be used as a p-type semiconductor and an antiferromagnetic material [18,19]. The distribution of cations on spinel  $\text{Co}_3\text{O}_4$  is shown to be  $\text{Co}^{2+}[\text{Co}_2^{3+}] \text{O}_4^{2-}$ : the cations inside parentheses are octahedral and those outside are tetrahe-

drally coordinated with oxygen ions. The complex is stable up to 800 °C and decomposes to  $\text{CoO}$  [14,20,21] above 900 °C.

A preview paper [9] reported that variation in the morphology of cobalt oxides with calcination and reduction pretreatments. Figlarz et al. found that the hexagonal  $\text{Co}_3\text{O}_4$  was obtained by the decomposition of  $\text{CoO}(\text{OH})$  at approximately 250 °C, based on analysis by X-ray diffraction (XRD) and selected-area electron diffraction (SAED). To understand the phase transformation and the composition of decomposed outlet gases, this work discusses the analysis of the thermogravimetry-mass spectrometry (TG-MS). Temperature-programmed decomposition (TPD), X-ray diffraction (XRD), infrared spectroscopy (IR), Raman spectroscopy and temperature-programmed reduction (TPR) are employed to characterize a series of cobalt oxides.

## 2. Experimental

### 2.1. Preparation of cobalt oxides

The as-prepared cobalt oxide (assigned as  $\text{CoO}_x$ ) with a high valence state of cobalt was synthesized by the precipitation–oxidation method in an aqueous solution. The precipitation process was carried out at 50 °C with 50 ml of 0.6 M  $\text{Co}(\text{NO}_3)_2 \cdot \text{H}_2\text{O}$  solution added drop by drop to 100 ml of 3.2 M NaOH solution; 100 ml of  $\text{H}_2\text{O}_2$  (50 wt%) was then introduced dropwisely under constant stirring. In order to avoid the contamination of chloride ion, the  $\text{H}_2\text{O}_2$  was chosen as an oxidizing agent instead of NaOCl [22]. The precipitate was then filtered, washed with deionized water and dried in

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an oven at 110 °C for 24 h. The dried  $\text{CoO}_x$  was ground and preserved in a desiccator as fresh samples. Further, other series of pure cobalt oxide were refined from the decomposition of  $\text{CoO}_x$  under nitrogen environment at different temperatures: 280, 450 and 950 °C, respectively (assigned as D-280, D-450 and D-950, respectively).

## 2.2. Characterization techniques

X-ray diffraction (XRD) measurements were performed using a Siemens D5000 diffractometer with  $\text{Cu K}\alpha 1$  radiation ( $\lambda = 1.5405 \text{ \AA}$ ) at 40 kV and 30 mA with a scanning speed in  $2\theta$  of  $2^\circ \text{ min}^{-1}$ . Using the XRD diffraction data, the crystallite sizes of cobalt oxides were able to be estimated using the Scherrer equation.

$$d = \frac{0.9\lambda}{B \cos \theta_B} \quad (1)$$

In this equation,  $d$  is the crystallite size (nm);  $\lambda$  the X-ray wavelength;  $B$  the full width at half maximum of the diffraction peak at  $\theta_B$ ;  $\theta_B$  is the diffraction angle.

Specific surface area measurements were carried out by using the Brunauer–Emmett–Teller (BET) method on a Micromeritics ASAP 2010 apparatus. Nitrogen adsorption isotherms at  $-196^\circ\text{C}$  were determined volumetrically. The catalysts were pre-outgassed at  $5 \times 10^{-5}$  Torr for 3 h at 110 °C. The surface area was determined from the nitrogen adsorption isotherm.

IR spectra of samples were obtained by a Bomem DA-8 spectrometer with a resolution of  $4 \text{ cm}^{-1}$  in the range of  $450\text{--}1000 \text{ cm}^{-1}$ . One milligram of each powder sample was diluted with 200 mg of vacuum-dried IR-grade KBr powder and subjected to a pressure of 10 tons. The measurements of Raman spectroscopy were recorded using a Nicolet Almega XR dispersive Raman spectrometer. The spectra were collected between  $300$  and  $900 \text{ cm}^{-1}$ , using a beam of diode laser (780 nm), with the sample exposed to the air under ambient conditions.

TPR of a series of cobalt oxides was performed using 10%  $\text{H}_2$  in Ar as the reducing gas. The flow rate of  $\text{H}_2/\text{Ar}$  was adjusted by mass flow controller under  $25 \text{ ml min}^{-1}$ . The cell was a quartz tube with an inner diameter 8 mm and 80 mg of the catalyst was mounted with quartz wool. The hydrogen consumption was monitored by a thermal conductivity detector (TCD) on raising the sample temperature from RT to 500 °C at a constant rate of  $5^\circ\text{C min}^{-1}$ . TPD profile of  $\text{CoO}_x$  sample was performed using He as carrier gas. The flow rate of He was adjusted to  $25 \text{ ml min}^{-1}$ . The oxygen desorption was monitored by a TCD on raising the sample temperature from RT to 1000 °C at a constant rate of  $10^\circ\text{C min}^{-1}$ .

On-line TG-MS analyses of the gases evolved from the sample were performed simultaneously using the STA-409CD with Skimmer coupled to a quadruple mass spectrometer QMA 400 (maximum 512 amu). The TG experiments were operated from RT to 1100 °C with heating rate of  $10^\circ\text{C min}^{-1}$  under a continuous flow of He ( $100 \text{ ml min}^{-1}$ ). A transfer line, specially designed to connect a vacuum pump in order to optimize the amount of evolved gas, transferred from the TG to the MS. A mass analysis was performed each 2 s recording mass fragments between 2 and 300 atomic mass

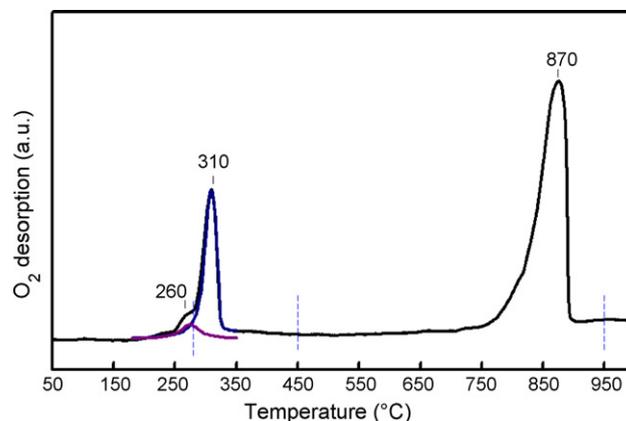


Fig. 1. TPD profile of  $\text{CoO}_x$ .

units (amu). The measurement of outlet gas via mass spectrometric intensities was plotted as a function of temperature that showed two well separated regions of de-volatilization.

## 3. Results and discussion

### 3.1. Structural analysis of cobalt oxides

Fig. 1 displays the TPD profile of the as-prepared  $\text{CoO}_x$  sample. Three peaks of oxygen desorption at different temperatures ( $T_d$ ) –  $D_1$  ( $T_d = 260^\circ\text{C}$ ),  $D_2$  ( $T_d = 310^\circ\text{C}$ ) and  $D_3$  ( $T_d = 870^\circ\text{C}$ ) – are observed. The sequential desorption of oxygen from  $\text{CoO}_x$  indicates that cobalt oxide with different oxidation states may be obtained by the appropriate thermal decomposition of  $\text{CoO}_x$  in a nitrogen environment.



Cobalt ions with an oxidation state of +3 [ $\text{CoO(OH)}$ ], +8/3 [ $\text{Co}_3\text{O}_4$ ] and +2 [ $\text{CoO}$ ] may be obtained from  $\text{CoO}_x$  by thermal decomposition at 280, 450 and 950 °C, respectively. Samples prepared in this way are designated herein as D-280, D-450 and D-950. Although  $D_1$  and  $D_2$  peaks in the spectrum are merged together, the desired pure sample of  $\text{CoO(OH)}$  is easily obtained by the simple thermal decomposition of  $\text{CoO}_x$  at 280 °C in the TPD system (see the characterizations below).

Fig. 2 shows XRD patterns of refined species of cobalt oxide. Columns 2–4 of Table 1 presents the species, crystal phase and particle size of these oxides. The diffraction profile of the D-280 sample (Fig. 2(a)) matches the JCPDS (PDF-74-1057) [23] file, identifying cobalt oxyhydroxide,  $\text{CoO(OH)}$ , with a hexagonal structure. The average particle size is small (10 nm, according to the peak width and the Scherrer equation). The D-450 sample (Fig. 2(b)) matches the JCPDS (PDF-76-1802) file identifying cobaltic oxide,  $\text{Co}_3\text{O}_4$ , with a spinel structure. The D-950 sample (Fig. 2(c)) matches the JCPDS (PDF-71-1178) file for cobaltous oxide,  $\text{CoO}$ , with a face-centered cubic (fcc) structure. The peak width in the D-950 pattern

Table 1

A series of the species of cobalt oxide for characterization

Samples	XRD		$S_{\text{BET}}$	FT-IR ( $\text{m}^2 \text{ g}^{-1}$ ) $\nu_{\text{Co-O}}$ ( $\text{cm}^{-1}$ )	Raman shift ( $\text{cm}^{-1}$ )	TPR (C)		
	Structure <sup>a</sup>	$d$ (nm) <sup>b</sup>				$\text{Co}_3\text{O}_4$	CoO	Co
CoOOH	Hexagonal	10	59	584	367, 482, 599, 809	240	295	500
$\text{Co}_3\text{O}_4$	Spinel	11	55	570, 661	482, 519, 621, 690	–	340	500
CoO	fcc	16	5	507	455, 675	–	–	550

<sup>a</sup> According to the data base of JCPDS2001.

<sup>b</sup> Particle size by Debye–Scherrer equation with  $\text{CoOOH}(1\ 1\ 1)$ ,  $\text{Co}_3\text{O}_4(3\ 1\ 1)$  and  $\text{CoO}(2\ 0\ 0)$  peak.

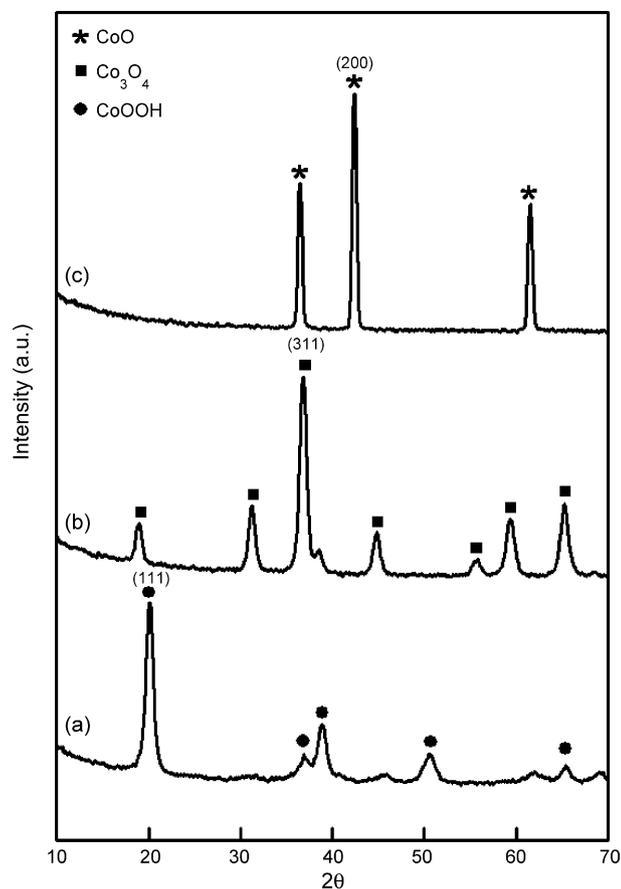


Fig. 2. XRD profiles of (a) D-280 (b) D-450 (c) D-950.

indicates substantial sintering of primary crystallites ( $d$  increases to 16 nm). Nitrogen adsorption isotherms on a series of cobalt oxides are obtained at  $-196^\circ\text{C}$ . They are similar across the adsorbents of interest. They are type II by Brunauer's classification. The decrease in the surface area (column 5 in Table 1) is associated with the sintering of D-950 sample.

The phase transformation revealed by XRD is accompanied by simultaneous variation in both IR and Raman spectra. Fig. 3 shows the IR absorption spectra of D-280, D-450 and D-950 samples. The IR spectra of D-280 and D-950 samples (Fig. 3(a) and (c)) display single bands ( $584$  and  $507\text{ cm}^{-1}$ ) that are likely to be associated with the cobalt ion in octahedral holes, meaning in an oxygen octahedral environment [24].

The variation may be caused by the difference between their structures (hexagonal vs. face-centered cubic). The IR spectrum of D-450 sample (Fig. 3(b)) displays two distinct bands that originate from the stretching vibrations of the metal-oxygen bonds [22,23,25]. The first band ( $\nu_1$ ) at  $570\text{ cm}^{-1}$  is associated with the  $\text{OB}_3$  vibration in the spinel lattice, where B denotes  $\text{Co}^{3+}$  in an octahedral hole. The second band ( $\nu_2$ ) at  $661\text{ cm}^{-1}$  is attributed to the  $\text{ABO}_3$  vibration, where A denotes the  $\text{Co}^{2+}$  in a tetrahedral hole.

Fig. 4 displays the Raman spectra of a series of the cobalt oxides formed by thermal decomposition. The D-280 sample has bands at  $367$ ,  $482$ ,  $599$  and  $809\text{ cm}^{-1}$  (Fig. 4(a)), which are assigned to  $\text{CoO}(\text{OH})$ . Following the higher-temperature treatments, different bands were observed at  $482$ ,  $519$ ,  $621$ ,  $690\text{ cm}^{-1}$  for D-450 (Fig. 4(b)) and  $455$ ,  $675\text{ cm}^{-1}$  for D-950 (Fig. 4(c)), probably because of the phase transformation under heat treatment. The prominent Raman peaks correspond to the  $E_g$  ( $482\text{ cm}^{-1}$ ),  $F_{2g}$  ( $519$  and  $621\text{ cm}^{-1}$ ),  $A_{1g}$  ( $690\text{ cm}^{-1}$ ) modes of the  $\text{Co}_3\text{O}_4$  crystalline phase (D-450) and are

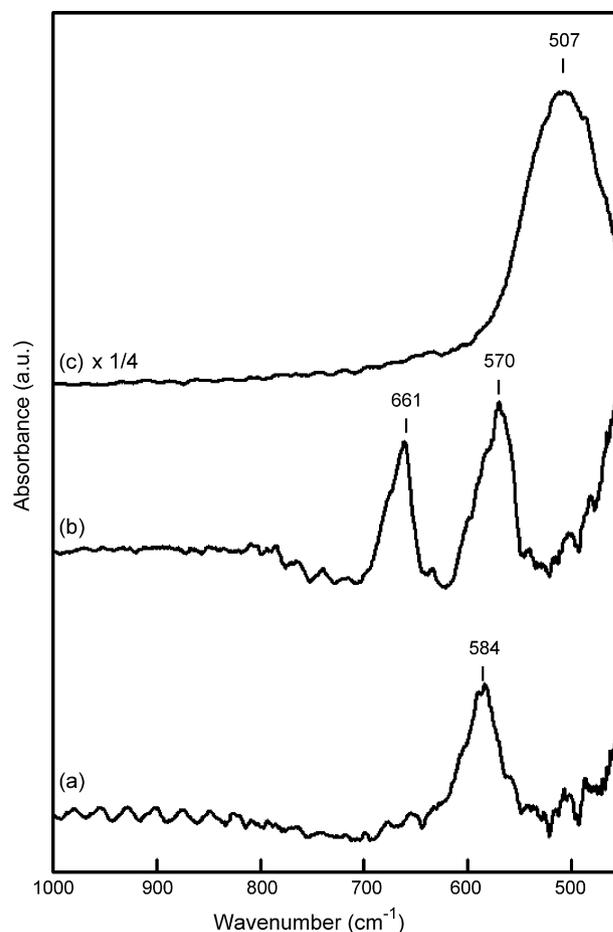


Fig. 3. FTIR spectra of (a) D-280 (b) D-450 (c) D-950.

consistent previous investigations [26,27]. Upon heat treatment at  $950^\circ\text{C}$ , the  $519$  and  $621\text{ cm}^{-1}$  peak disappeared and broad bands at  $455$  and  $675\text{ cm}^{-1}$  appeared, because of the formation of cubic  $\text{CoO}$ . These are shifted from the positions of the main peaks,  $672$  and  $468\text{ cm}^{-1}$ , in previous studies of bulk  $\text{CoO}$  [28–30], shifted because of the nano-sized effect. This result confirms that the crystalline structure of the series of cobalt oxides can be identified by Raman spectroscopy.

### 3.2. Reductive behavior of cobalt oxides

Fig. 5 compares the reduction behavior of the series of cobalt oxides. The D-280 sample (Fig. 5(a)) yields three sequent signals at  $215$ ,  $260$  and  $350^\circ\text{C}$ . According to our previous study [2], we have stopped each TPR step (under three reduction temperature) to record the XRD pattern of the sample and then to have a definitive knowledge of the phase formed after each reduction peak (from  $\text{CoO}(\text{OH})$  to  $\text{Co}_3\text{O}_4$ , then to  $\text{CoO}$ ). So, we suggest that the  $\text{CoO}(\text{OH})$  (D-280) is initially reduced to  $\text{Co}_3\text{O}_4$ , and then further reduced to  $\text{CoO}$  and  $\text{Co}$  metal. The following equations are designated.

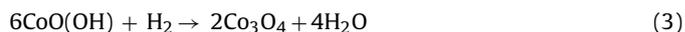


Fig. 5(b) shows the TPR profile of D-450 sample; it includes only two reductive singles at  $300$  and  $367^\circ\text{C}$ . Sexton et al. [31] found that the reduction profile of  $\text{Co}_3\text{O}_4$  includes a low-temperature peak below  $300^\circ\text{C}$  and a high-temperature peak at approximately

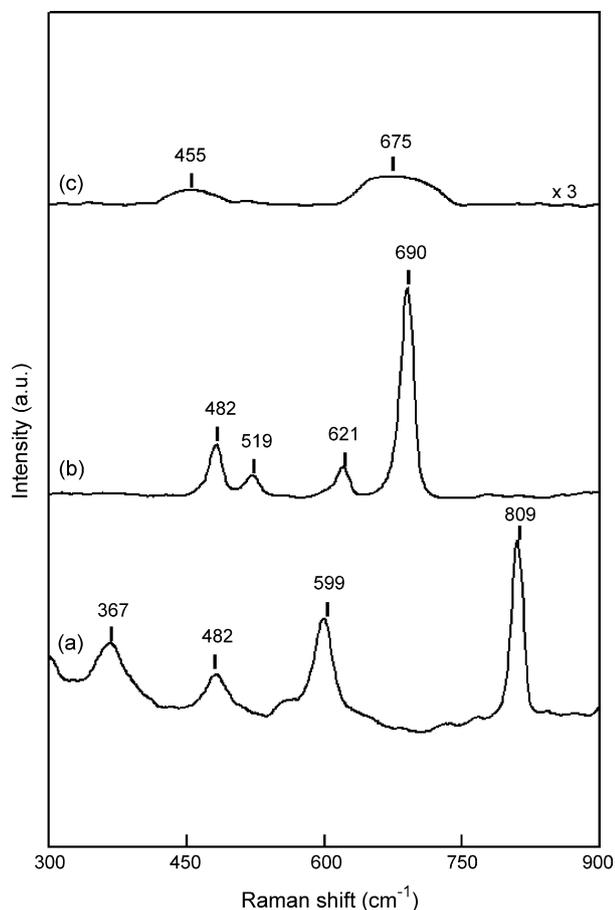


Fig. 4. Raman spectra of (a) D-280 (b) D-450 (c) D-950.

500 °C. According to the literature [32–35], the low-temperature peak is associated with the reduction of  $\text{Co}^{3+}$  ions to  $\text{Co}^{2+}$  (Eq. (4)) with the subsequent structural change to CoO. The second higher temperature peak is due to the reduction of CoO to metallic cobalt (Eq. (5)). A comparison with the D-280 sample, and in particular, the disappearance of the peak at around the lowest temperature indicates that the reductive behavior is in good agreement with the proposed  $\text{Co}_3\text{O}_4$ . The TPR profile of the D-950 sample (Fig. 5(c)) includes only a single peak, which is assigned unambiguously to the direct reduction of CoO to cobalt metal.

### 3.3. Analysis of evolved gas of cobalt oxides

To confirm the phase transformation of  $\text{CoO}(\text{OH})$  to  $\text{Co}_3\text{O}_4$  and CoO under heat treatment, coupled quantitative and qualitative analysis by TG-MS was performed to prove the structural identifications in Section 3.1. Fig. 6 presents the TG-MS profile of the D-280 sample. The thermogravimetric plot presents two marked weight losses at about 280 and 850 °C. Column 4 in Table 2 shows that these are 12% and 20%, respectively. The coupled TG-MS analysis reveals that the evolved gases are oxygen ( $\text{O}_2$ ,  $m/z=32$ ) and steam ( $\text{H}_2\text{O}$ ,  $m/z=18$ ). The red curve assigned to  $\text{O}_2$  has a small first peak and a large second peak. The blue curve assigned to  $\text{H}_2\text{O}$  is related to the crystal structure. Comparing the theoretical weight loss of the proposed reactions (columns 2 and 3 in Table 2) with the experimental results demonstrates the simultaneous desorption of  $\text{O}_2$  and  $\text{H}_2\text{O}$  from  $\text{CoO}(\text{OH})$  into  $\text{Co}_3\text{O}_4$  at around 280 °C.

The second weight loss, confirmed by MS analysis, shows the decomposition of  $\text{Co}_3\text{O}_4$  to CoO at around 850 °C. Therefore, the

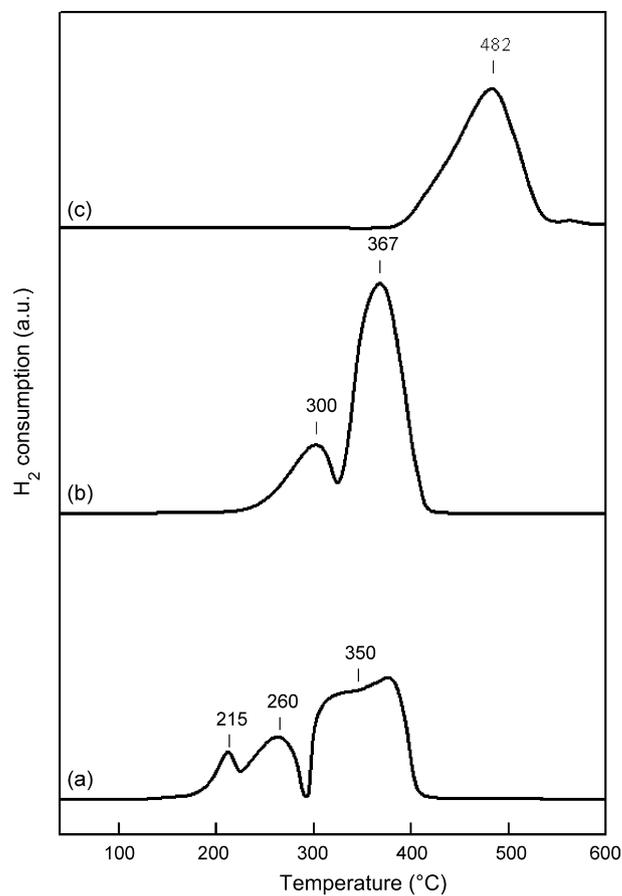


Fig. 5. TPR profiles of (a) D-280 (b) D-450 (c) D-950.

D-280 sample can be decomposed into  $\text{Co}_3\text{O}_4$  and then CoO under heat treatment.

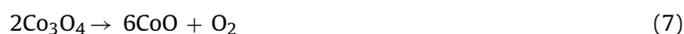
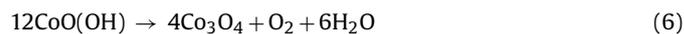


Fig. 7 shows the TG-MS profile of the D-450 sample. At under 300 °C, the tardy weight loss is associated with desorption of adsorbed water on the surface of  $\text{Co}_3\text{O}_4$ . The red curve is assigned

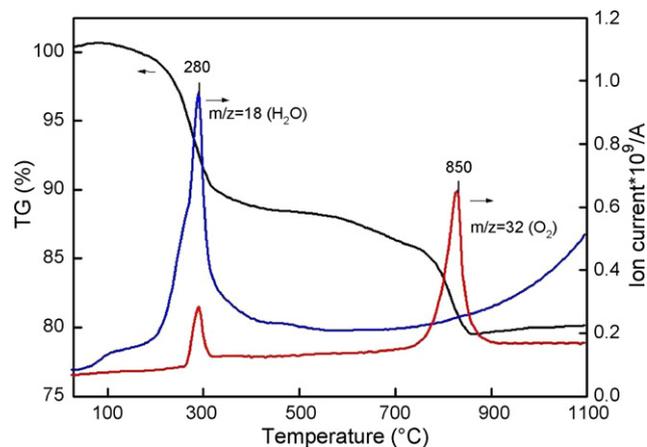


Fig. 6. Analysis of the outlet gases from the decomposition of D-280 by TG-MS (heating rate of  $10^\circ\text{C min}^{-1}$  under He):  $\text{H}_2\text{O}$  (blue) and  $\text{O}_2$  (red). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

**Table 2**  
Experimental and theorization of a series of cobalt oxide for TG-MS

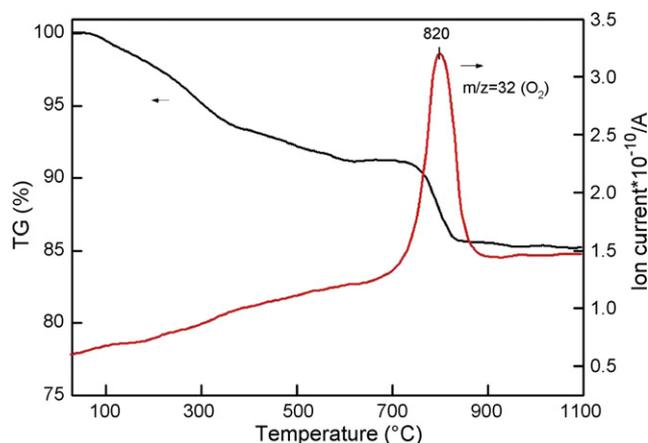
Sample	Reaction process	Weight loss (%) <sup>a</sup>		Outlet gas (C) <sup>b</sup>	
		Theoretical	Experimental	H <sub>2</sub> O	O <sub>2</sub>
CoOOH <sup>c</sup>	12CoO(OH) → 4Co <sub>3</sub> O <sub>4</sub> + O <sub>2</sub> + 6H <sub>2</sub> O	13	12	300	920
	2Co <sub>3</sub> O <sub>4</sub> → 6CoO + O <sub>2</sub>	19	20		
Co <sub>3</sub> O <sub>4</sub> <sup>c</sup>	2Co <sub>3</sub> O <sub>4</sub> → 6CoO + O <sub>2</sub>	7	8	–	820
CoO <sup>d</sup>	6CoO + O <sub>2</sub> $\xrightarrow{A}$ 2Co <sub>3</sub> O <sub>4</sub> $\xrightarrow{B}$ 6CoO + O <sub>2</sub>	A:7 B:-7	A:7 B:-8	–	920

<sup>a</sup> Measurement of TG.

<sup>b</sup> Measurement of mass analyzer.

<sup>c</sup> By TG-MS flow He.

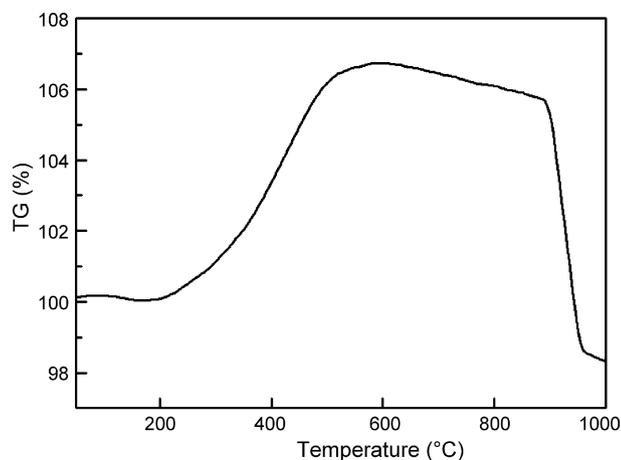
<sup>d</sup> By TG-MS flow air.



**Fig. 7.** Analysis of the outlet gases from the decomposition of D-450 by TG-MS (heating rate of 10 °C min<sup>-1</sup> under He): O<sub>2</sub> (red). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of the article.)

to O<sub>2</sub> at 820 °C. The weight loss of 8% is caused mainly by the decomposition of Co<sub>3</sub>O<sub>4</sub> to CoO, according to Eq. (7).

Fig. 8 shows the TG profile of the D-950 sample in air. The weight increase of 7% involves mainly the oxidation of CoO to Co<sub>3</sub>O<sub>4</sub> as the temperature is increased from 250 to 900 °C, followed by the desorption of oxygen to CoO at above 900 °C. Comparing the theoretical weight variations of the proposed reaction (columns 2 and 3 in Table 2) with the experimental results demonstrated that CoO can undergo oxidation–reduction in air, according to Eq. (8).



**Fig. 8.** TG profile of D-950 at a heating rate of 10 °C min<sup>-1</sup> under air.

## 4. Conclusions

A convenient procedure for preparing a series of pure cobalt oxides is presented; it involves the decomposition of CoO<sub>x</sub> in a nitrogen environment at different temperatures.

TG-MS, XRD, Raman, FTIR and TPR analyses are demonstrated to be useful in identifying the series of cobalt oxides. The D-280 sample [CoO(OH)] has a hexagonal structure that gives rise to Co<sub>3</sub>O<sub>4</sub> and CoO in helium at 280 and 850 °C, respectively. The D-450 sample [Co<sub>3</sub>O<sub>4</sub>] has a spinel structure that gives rise to CoO in helium at 820 °C. The D-950 sample [CoO] has a faced-centered cubic structure that can be oxidized to Co<sub>3</sub>O<sub>4</sub> at 200–600 °C and decomposed to CoO at 850–1000 °C in air, respectively. The preparation of materials and the analysis of their thermal behavior is a convenient way to control and determining the purity of cobalt oxide.

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