

# USING SiO<sub>2</sub> NANOPARTICLES TO EFFICIENTLY ENHANCE LIGHT EMISSION FROM METAL-OXIDE-SILICON TUNNELING DIODES ON Si

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## ABSTRACT

Silicon dioxide nanoparticles are used as oxide layer in metal-oxide-silicon tunneling diodes. With its non-uniformity in thickness, tunneling current is concentrated in carrier accumulation region. Both electron and hole can be confined to enhance radiative recombination rate. Electroluminescence at silicon band edge (1.1 $\mu$ m) with external quantum efficiency  $1.5 \times 10^{-4}$  has been achieved. KOH wet etch also contributes to improvement on efficiency by removing nonradiative recombination center at surface. Frequency response is exploited as tool for extraction radiative and nonradiative recombination rate.

## 1. INTRODUCTION

Silicon is the most important semiconductor material for the electronics industry. However, its importance in optoelectronics is not prominent because indirect-bandgap nature of Si makes it inefficient as a light source. Therefore optoelectronic integration on the silicon chip is a challenging issue. Many efforts have been devoted for decades to making silicon useful for optoelectronics. These efforts include optical waveguides [1] or micro-optics [2] on Si, as well as converting silicon into light-emitting material such porous-silicon-based devices [3-5], nanocrystalline Si [6,7], Si<sup>+</sup>-implanted SiO<sub>2</sub>[8,9], deposition of polymer upon Si substrates[10], growth of GaN upon Si substrates[11,12], and so on.

In this work we report that using SiO<sub>2</sub> nanoparticles as oxide layer in metal-oxide-silicon (MOS) tunneling diode and KOH wet etch to improve surface condition, external quantum efficiency of electroluminescence (EL) at silicon bandgap edge (wavelength 1.1 $\mu$ m) can be enhanced to  $1.5 \times 10^{-4}$ . In silicon, electrons and holes cannot recombine efficiently due to momentum mismatch. To overcome this difficulty, we use SiO<sub>2</sub> nanoparticles to generate nano-structural confinement of electrons and holes. Traditional oxide layer is replaced by SiO<sub>2</sub> nanoparticles (particle size 8~12nm) [13]. With its non-uniformity in oxide thickness, two major effects take place. First, in n-type silicon, holes tunneling through thin region of oxide layer are orders of magnitudes more than those through thick region and this leads to concentration of injected minority carriers (holes). Second, band bending toward the Si/SiO<sub>2</sub> interface also depends on oxide thickness. This leads to local concentration of accumulated majority carriers (electrons). As a result, excitons or electron-hole pairs are more easily formed around thin oxide region than in bulk silicon and radiative recombination rate increases.

Experiments (frequency response as well as varying efficiency) show that nano-structures in the oxide layer could enhance radiative recombination rate to comparable with Schottky-Read-Hall (SRH) nonradiative recombination rate. This indicates that nonradiative recombination is not the only dominant factor any more, and using SiO<sub>2</sub> nanoparticles can enhance silicon EL for orders of magnitude.

To further promote efficiency, KOH wet etch [14] is used to improve surface quality of silicon. In commercially available lightly-doped silicon ( $10^{15} \text{cm}^{-3}$ ), bulk minority lifetime can be as long as lms. Nonradiative recombination rate is reduced to about  $10^3 \text{sec}^{-1}$  inside bulk silicon, which implies radiative recombination rate can be possibly larger than nonradiative one. However, nonradiative combination through surface states is still a severe problem on prohibiting light emission. Experiments show KOH wet etch can significantly increase EL intensity because KOH solution starts etch surface defects, which are centers of nonradiative recombination.

By using KOH wet etch on silicon surface and SiO<sub>2</sub> nanoparticles as oxide layer, we are able to fabricate MOS light emitting diodes with efficient light emission. The measured light-current (L-I) curve shows that external EL quantum efficiency rapidly increases and reaches  $1.5 \times 10^{-4}$  for operation current between 20mA to 30mA. Output power can be 10 $\mu$ W at injection current of 100mA (or current density 12.7A/cm<sup>2</sup>). This power is measured only from periphery of top metal pad. If total EL is included, quantum efficiency should be more than  $10^{-3}$ .

## 2. DEVICE FABRICATION

The processing steps are as follows. First, native oxide on silicon wafer is removed by buffered oxide etch (BOE) and KOH solution (45%) is subsequently applied to improve surface quality. Prepared SiO<sub>2</sub> nanoparticles

solution with feature size 8–11 nm is spin-coated and baked in oven. The thickness of coated oxide layer is about 150 nm. A thick layer of Al was evaporated on wafer backside to form ohmic contact. Conductive silver paint is directly applied on top of the nanoparticles to hold gold wire for top contact. Fig.1 shows scanning-electron microscope photograph of cross-sectional view and schematic of device (inset).

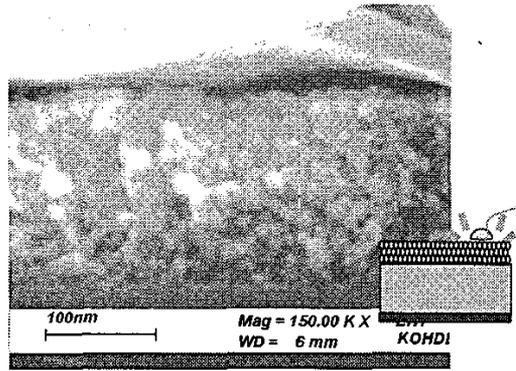


Fig. 1 Schematic of MOS tunneling diode with SiO<sub>2</sub> nanoparticles as oxide layer and SEM photo of cross-sectional view.

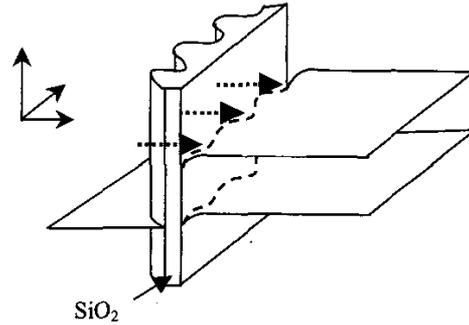


Fig. 2 Two-dimensional band diagram of electroluminescent MOS tunneling diode on n-type silicon with non-uniform oxide.

In analysis of this MOS tunneling diode with non-uniform oxide, Fig. 2 shows two-dimensional band diagram. Dashed arrows point to thin oxide regions, where most holes tunnel into silicon under positive bias applied to metal. In silicon under thin oxide, electrons accumulate heavily with more surface bending. Spatial confinement of electrons and holes in this nanostructure improves radiative recombination rate. Because SiO<sub>2</sub> nanoparticles are not tightly packed, there are a lot of hollow spaces among these nanoparticles. After applying conductive silver paint on top of oxide, silver will migrate under appropriate electric field and arrive at silicon surface until large tunneling current is achieved.

Polish damage is common nonradiative recombination centers for commercially available prime grade silicon wafer. Wet etching solution KOH is used to improve surface quality of silicon wafer and contributes to higher efficiency in emitting light at Si bandgap edge. The EL spectrum with origin of phonon-emission exciton recombination [15] is shown in the Fig. 3. Commercial 45% KOH is convenient for use. The etching rate of the 45% KOH solution at 35°C is about 1 nm/sec. Fig. 4 shows the experiments that KOH etching silicon wafer for 20 seconds can increase the EL intensity for more than 10 times in comparison with the devices without KOH etching (0 second etching time).

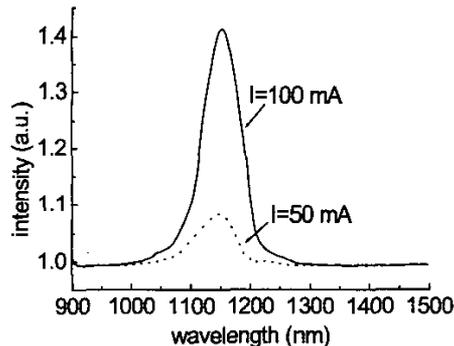


Fig. 3 The EL spectra correspond to Si bandgap edge.

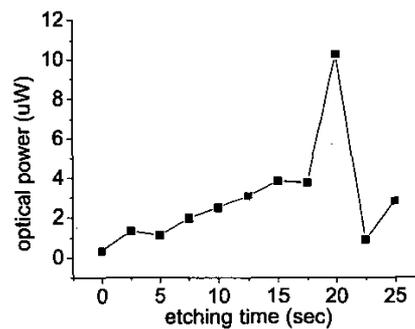


Fig. 4 Variation of EL power with KOH etching time

The L-I curve in Fig. 5 shows that EL external quantum efficiency rapidly increases and reaches  $1.5 \times 10^{-4}$  for operation current between 20 to 30 mA and slightly drops to  $10^{-4}$  due to thermal effect. The EL power could be  $10 \mu\text{W}$  at the injection current of 100 mA (current density  $12.7 \text{ A/cm}^2$ ). The power is measured only from the periphery of the silver paint pad. Because top contact pad would obstruct a portion of light generated from the

device, actual light power might be more than  $10 \mu\text{W}$ . If total EL power is included, quantum efficiency will reach  $10^{-3}$ .

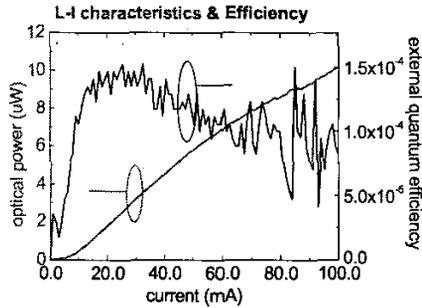


Fig. 5 Output power & external quantum efficiency vs. injection current probably exceed  $10^{-3}$ .

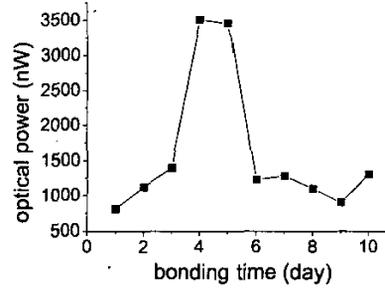


Fig. 6 Variation of EL power with air exposure days.

As shown in Fig. 6, EL intensity increases for first few days and then decreases. The reason for increasing efficiency is formation of native oxide of silicon surface to passivate surface states and to prevent short contact of silver and silicon. Nonradiative recombination is therefore reduced. For overgrown native oxide, high electric field is needed for the same level of tunneling current. Certain region of oxide will breakdown with metal punch-thru. Light efficiency reduces and resembles that of first few days.

### 3. CHARACTERISTICS BY FREQUENCY RESPONSE

Frequency response uses small signal current as less-destructive excitation and permits measurement of carrier lifetime under low level bias condition ( $< 10\text{A}/\text{cm}^2$ ). In analysis of carrier lifetime, both radiative and nonradiative recombination rate and its coefficient can be determined by relation between lifetime and current bias. The physical reason for lifetime-bias relation is attributed to extension of accumulation region near surface into silicon bulk region. In EL MOSTD, light is proportional to current in static state condition. With modulated current, response time of light is limited by carrier recombination rate. To be specific, the rate equation describing minority dynamics is  $\frac{dp}{dt} = \frac{\eta J}{ed} - (A + Bn)p$ . Minority carrier recombination time is defined as  $\tau = 1/(A + Bn)$ .

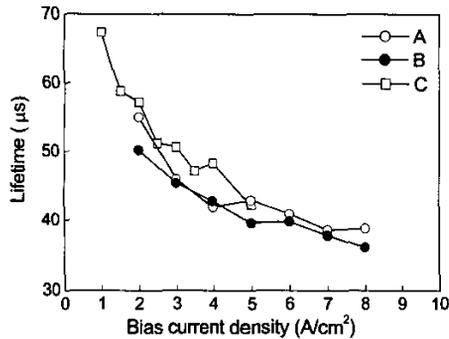


Fig. 7 Measured lifetime of EL MOSTD for different fabrication condition: A, B) normal C) with KOH removal of polish damage.

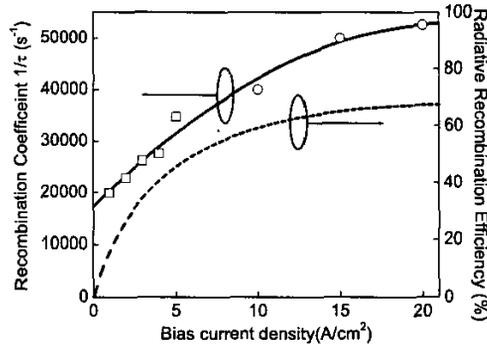


Fig. 8 Fitted recombination coefficient and theoretical recombination efficiency of EL MOSTD

Fig. 7 shows the relation between measured lifetime from frequency response and bias current. Device A and B are MOSTD fabricated by application of 120nm thick layer of  $\text{SiO}_2$  nanoparticle. Device C is fabricated in the same way except being pre-treated with KOH wet etch. All devices have metal pad area  $0.01\text{cm}^2$ . The external light emission efficiency  $\eta_{\text{ext}}$  is  $5 \times 10^{-6}$ ,  $2 \times 10^{-6}$  and  $10^{-5}$  (W/A) respectively. With the same light extraction efficiency  $\eta_c$  and minority carrier injection efficiency  $\eta_{\text{inj}}$ , internal or radiative recombination efficiency  $\eta_{\text{rec}}$  is proportional to external efficiency because  $\eta_{\text{ext}} = \eta_c \eta_{\text{inj}} \eta_{\text{rec}}$ . Devices with longer lifetime with low level bias current have better efficiency and have quite the same lifetime with high level bias current. This suggests that initially lifetime be affected by nonradiative recombination and lowering effect comes from

increasing radiative recombination by increasing carrier density.

Inverse of lifetime, or recombination coefficient  $\tau^{-1} = A + Bn$ , under certain level of injection current density  $J$ , reveals simple relation with carrier density  $n$ . However, it lacks theoretical explanation of complex relation between  $n$  and  $J$ . Observation in Fig. 8 allows us to empirically approach  $n$ - $J$  relation by  $n = \beta J - \gamma J^2$ . This saturation behavior of carrier density comes from physical consideration that increasing injection current at high level only extends accumulation depth rather than increases  $n$ . Carrier lifetime is related to injection current density by  $\tau^{-1} = A + B\beta J - B\gamma J^2$  as shown by solid line in Fig. 8.

This measurement fitting permits identification of individual contribution of radiative and nonradiative recombination rate. Fitted  $A$  is 17185 ( $\text{sec}^{-1}$ ),  $B\beta$  is 320 ( $\text{sec}^{-1}\text{A}^{-1}\text{cm}^2$ ) and  $B\gamma$  is 0.718 ( $\text{sec}^{-1}\text{A}^{-1}\text{cm}^2$ ). However, only nonradiative recombination coefficient can be independently extracted. With typical value of  $B$  coefficient as  $10^{-15} \text{cm}^3/\text{sec}$  [6], majority carrier density  $n$  is in order of  $10^{19}\sim 10^{20} \text{cm}^{-3}$ .

Ratio of radiative recombination rate to overall recombination rate, or radiative recombination efficiency  $\eta_{\text{rec}} = Bn/(A + Bn)$  can be obtained, as shown by dash line in Fig. 8. This ratio reaches more than 50% at injection current density  $5\text{A}/\text{cm}^2$ , indicating that radiative recombination is more efficient than nonradiative recombination. This high recombination efficiency comes from selection of high quality silicon wafer for fabrication of EL MOSTD and using  $\text{SiO}_2$  nanoparticles to provide improved carrier confinement to increase carrier density. Although internal efficiency is high, external efficiency is limited by low light extraction efficiency due to metal pad and low minority injection efficiency.

With frequency response analysis of radiative and nonradiative recombination rate, it is confirmed that internal efficiency of our light emitting diode is very high.

#### 4. CONCLUSION

We have demonstrated that use of  $\text{SiO}_2$  nanoparticles and KOH wet etch can improve electroluminescence of MOS tunneling diode based on silicon. The possibility of efficient EL from MOS tunneling diodes gives great promise on integrated optoelectronics of silicon.

#### REFERENCES

- [1] N. Takato, K. Jinguji, M. Yasu, H. Toba and M. Kawachi, *IEEE J. Lightwave Technol.* **6**, 1003 (1988)
- [2] M. C. Wu, L. Y. Lin, S. S. Lee and K. S. Pister, *Sensors Actuators A* **50**, 127 (1995)
- [3] L. T. Canham, *Appl. Phys. Lett.* **57**, 1046 (1990)
- [4] N. Koshida and H. Koyama, *Appl. Phys. Lett.* **60**, 347 (1992)
- [5] S. Lazarouk, P. Jaguiro, S. Katsouba, G. Masini, S. La Monica, G. Maiello and F. Ferrari, *Appl. Phys. Lett.* **68** (1996)
- [6] S. Schuppler et al, *Phys. Rev. B* **52**, 4910.
- [7] D. B. Geohegan, A. A. Puretzky, G. Duscher and S. J. Pennycook, *Appl. Phys. Lett.* **68**, 2410 (1998)
- [8] L. S. Liao, X. M. Bao, X. Q. Zheng, N. S. Lo and N. S. Min, *Appl. Phys. Lett.* **68**, 850 (1996)
- [9] W. Skorupa, R. A. Yankov, I. E. Tyschenko, H. Frob, T. Bohme, and K. Leo, *Appl. Phys. Lett.* **68**, 2410 (1996)
- [10] D. R. Baigent, R. N. Marks, N. C. Greenham, R. H. Friends, S.C. Moratti, and A. B. Holms, *Appl. Phys. Lett.* **65**, 2636 (1994)
- [11] R. Birkhahn and A. J. Steckl, *Appl. Phys. Lett.* **73**, 1700 (1998)
- [12] M. Garter, J. Scofield, R. Birkhahn and A. J. Steckl, *Appl. Phys. Lett.* **74**, 182 (1999)
- [13] C. F. Lin, P. F. Chung, M. J. Chen and W. F. Su, *Optics Lett.* **27**, 713 (2002)
- [14] H. Seidel et al., *J. Electrochem. Soc.* **137**, 3612 (1990)
- [15] M. J. Chen, E. Z. Liang, S. W. Chang and C. F. Lin, *J. Appl. Phys.* **90**, 789(2001)