

# Available online at www.sciencedirect.com



Thin Solid Films 511-512 (2006) 199 - 202



# Novel poly(3-methylthiophene)-TiO<sub>2</sub> hybrid materials for photovoltaic cells

Yi-Jun Lin a, Leeyih Wang b,c,\*, Wen-Yen Chiu a,c

- <sup>a</sup> Department of Chemical Engineering, National Taiwan University, Taipei, Taiwan
- <sup>b</sup> Center for Condensed Matter Sciences, National Taiwan University, Taipei, Taiwan
- <sup>c</sup> Institute of Polymer Science and Engineering, National Taiwan University, Taipei, Taiwan

Available online 25 January 2006

### Abstract

Organic-inorganic hybrid materials based on poly(3-methylthiophene) (P3MeT) and  $TiO_2$  were developed and investigated. The  $TiO_2$  nanoparticles were deposited on ITO-coated glass by spin coating, and characterized with scanning electron microscopy (SEM), atomic force microscopy (AFM), X-ray diffraction (XRD) and Brunauer-Emmett-Teller (BET) equation. The P3MeT films were then electrodeposited onto the porous  $TiO_2$  film for junction formation. Instruments of AFM and electron spectroscopy for chemical analysis (ESCA) were utilized to examine the surface morphology and chemical composition of the P3MeT/ $TiO_2$  hybrids, respectively. The performance of the photovoltaic cell fabricated from as-prepared donor/acceptor hybrid was evaluated. © 2005 Elsevier B.V. All rights reserved.

Keywords: Electrochemical polymerization; Sol-gel methods; Solar cells

### 1. Introduction

Photovoltaic cells containing organic semiconductors have attracted considerable attention [1-3] because their electrical and optical properties can be varied widely, and generally they can be fabricated more easily than inorganic photovoltaic cells. However, the use of organic molecules or conducting polymers to fabricate photovoltaic cells remains impractical because of the small overlap of the electronic transition absorption spectrum of organic materials with the solar spectrum and ineffective charge transport to electrodes. Since the diffusion range of excitons in conjugated polymers typically ranges between 5-15 nm [4-7], a large interface area is required between the electron donor and electron acceptor to achieve a high efficiency of charge separation.

This work describes a new route for preparing organic—inorganic hybrid as a photoactive material in ultrathin photovoltaic device using an electrochemical polymerization technique. Because of the higher accessibility to small pores for thiophene monomer than for the polythiophene (PTh) chain, the interface area between the PTh phase (electron donor) and

E-mail address: leewang@ntu.edu.tw (L. Wang).

TiO<sub>2</sub> phase (electron acceptor) is expected to increase. Moreover, the subject method does not require the polymer to be soluble in the solvent, since monomers without solubilizing side chains or with very short side chains can be used. In this study, we chose nanoporous TiO<sub>2</sub> as the inorganic semiconductor and 3-methylthiophene as the monomer for electropolymerization. Polythiophene [8] was selected because of its good environmental stability and high hole-mobility.

# 2. Experimental

# 2.1. Synthesis and characterization of nanoporous titania film

Titanium isopropoxide, 2-propanol and nitric acid were purchased from Acros and used without further purification. TiO<sub>2</sub> colloid solutions were prepared by sol-gel reaction of titanium isopropoxide, Ti(OCH(CH<sub>3</sub>)<sub>2</sub>)<sub>4</sub>, as follows: under a stream of dry nitrogen, 25 mL of Ti(OCH(CH<sub>3</sub>)<sub>2</sub>)<sub>4</sub> was added to a dropping funnel containing 4 mL of 2-propanol. The mixture was added to 150 mL of deionized water over 10 min with vigorous stirring. Within 10 min of the alkoxide addition, 1.14 mL of 65% nitric acid was further added to the system. The reaction was continued for 8 h at 80 °C. The resulting sol was then concentrated under vacuum at room temperature until the TiO<sub>2</sub> concentration was about 80 gL<sup>-1</sup>. Finally, two drops of nonionic surfactant, Triton-X 100, were added to the

<sup>\*</sup> Corresponding author. Center for Condensed Matter Sciences, National Taiwan University, Taipei, Taiwan. Tel.: +886 2 3366 5276; fax: +886 2 2363 5404.

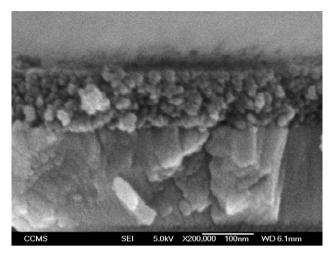


Fig. 1. SEM image of the cross section of a sintered TiO<sub>2</sub> film.

solution and the solution then was stirred for several hours to enhance the colloidal stability and size uniformity of TiO<sub>2</sub>.

Nanoporous TiO2 films were formed on ITO coated glass through spin coating at 2500 rpm. The films were heated to 450 °C at a heating rate of 5 °C min<sup>-1</sup> and held at 450 °C for 1 h. The morphology of the TiO2 films was examined by scanning electron microscopy (JOEL). Surface area of TiO<sub>2</sub> nanoparticles was determined from Brunauer-Emmett-Teller (BET) equation. X-ray diffraction measurements (Philips, X'Pert PRO) with monochromatic CuK<sub>\alpha</sub> radiation were taken to examine the crystal structure of the particles. Film thickness was measured on an ellipsometer (GAERTNER, L116D). The typical film thickness following calcinations was approximately 70 nm. XRD and ellipsometry measurements were conducted on films prepared on silicon wafers. Photoluminescence spectra were taken using a Fluorolog-Tau-3 spectrometer (Jobin Yvon) by photoexcitation at 480 nm. The depth profile of elemental compositions in the P3MeT/TiO2 hybrid was examined by electron spectroscopy for chemical analysis (ESCA) technique, which were performed on a VG Scientific ESCALAB 250 spectrometer under a Mg Kα (1253.6 eV) radiation source. Sample etching was undertaken in situ using a VG Scientific ESCALAB 250 EX05 argon ion etcher with an emission current of 15 mA. The etched crater was  $2 \times 5$  mm<sup>2</sup>.

# 2.2. Electropolymerization of 3-MeT into nanoporous titania films

Electropolymerization of 3-methylthiophene in nanoporous titania films was carried out in acetonitrile using  $Bu_4^n NBF_4$  (Aldrich) as the electrolyte. Acetonitrile was distilled before use, while other chemicals were used as received. The monomer and electrolyte concentrations were 0.1 and 0.02 M, respectively. The electropolymerization was controlled by an Electrochemical Analyzer (CH Instrument, Model 614A) using platinum and Ag/AgCl as the counter electrode and reference electrode, respectively. During the electropolymerization, a constant potential of 2.0 V was applied for several minutes to establish a highly  $BF_4^-$  doped P3MeT layer on and inside the porous  $TiO_2$  matrix. The samples were undoped at -0.8 V for

5 min then rinsed with fresh acetonitrile. The overgrown polymer layers were removed mechanically.

# 2.3. Fabrication of photovoltaic cells

Photovoltaic cells were fabricated by spin-coating a thin layer of poly(3-hexylthiophene) on the top of P3MeT/TiO<sub>2</sub>/ITO prepared above to avoid direct contact of the TiO<sub>2</sub> and top electrode, followed by the evaporation deposition of a 60 nm thick of Al electrode under vacuum. The effective cell area was adjusted to about 0.25 cm<sup>2</sup>. The measurement of the I-V characteristic of the cell was conducted while the cell was illuminated by an Xe lamp with intensity of 88 mW/cm<sup>2</sup>.

### 3. Results and discussion

This study produced nanoporous TiO<sub>2</sub> films via the sol-gel method using titanium isopropoxide as precursor, and then calcining the films at 450 °C. Fig. 1 illustrates the SEM cross-section image of a sintered TiO<sub>2</sub> film deposited on conducting glass substrate. The figure reveals that the TiO<sub>2</sub> film comprised a three-dimensional network of interconnected particles with an average size of approximately 14 nm. The BET surface area of the TiO<sub>2</sub> powers was about 89.45 m<sup>2</sup>/g. Fig. 2 displays X-ray diffraction pattern of the TiO<sub>2</sub> powders. Nanoparticle size was determined based on the diffraction peak by the use of Scherrer's equation [9]. This approach calculated that the crystallite size of the TiO<sub>2</sub> particle was approximately 9.8 nm.

Fig. 3 illustrates the surface morphologies of TiO<sub>2</sub> and P3MeT/TiO<sub>2</sub> films, as determined using atomic force microscope in the tapping mode. From the AFM images, the nanostructured TiO<sub>2</sub> in Fig. 3a comprised interconnected and nanometer sized TiO<sub>2</sub> particles and pores. The TiO<sub>2</sub> particles in the film have been observed to be uniform (root mean square surface roughness around 2.0 nm) with average diameter 18 nm. Fig. 3b shows the measurements performed following the deposition of the in situ formed P3MeT on the nanoporous TiO<sub>2</sub>-coated ITO glass. The surface roughness is significantly higher (rms roughness of 63.0 nm) for the composite films compared to the parent TiO<sub>2</sub> film. Comparison of the surface

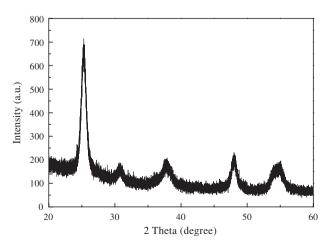


Fig. 2. X-ray diffraction pattern of TiO2 nanoparticles.

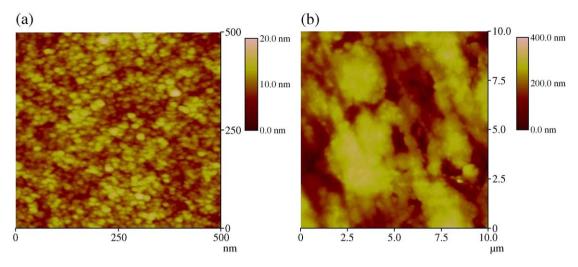


Fig. 3. AFM images of (a) TiO<sub>2</sub> and (b) P3MeT/TiO<sub>2</sub> film.

morphologies of the  $TiO_2$  layer with P3MeT coated  $TiO_2$  layers from the AFM pictures in Fig. 3 reveals that the surface of the  $TiO_2$  network was covered by the polymer layer.

Fig. 4 displays the absorption spectra of P3MeT/ITO and P3MeT/TiO<sub>2</sub>/ITO. Compared to P3MeT deposited on ITO, the maximum absorption of P3MeT grown inside nanoporous TiO<sub>2</sub> is blue-shifted, which indicates that the P3MeTs synthesized from the TiO<sub>2</sub>/ITO substrate possessed a smaller average conjugation length than those prepared directly from ITO substrate. This phenomenon may result from low molecular weight and twist chain conformation of P3MeT.

Electron spectroscopy for chemical analysis (ESCA) was employed to elucidate the P3MeT distribution in the depth profile, by examining the elemental compositions of the composite film. Fig. 5a displays the ESCA spectra of the surface of the P3MeT/TiO<sub>2</sub> film prepared by the electropolymerization of 3-methyl thiophene and the removal of the top layer of pure P3MeT, as specified in the Experimental section. The signals at 459.0, 530.4, 285.0 and 164.0 eV correspond to the binding energies of Ti(2p<sub>3/2</sub>), O(1s), C(1s) and S(2p<sub>3/2</sub>) [10–12], respectively, proving the coexistence of TiO<sub>2</sub> and P3MeT in the thus-synthesized film. Fig. 5b depicts the sulfur

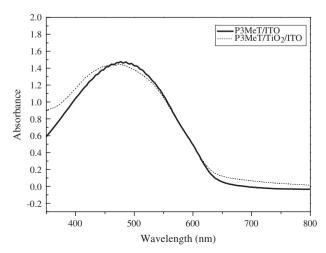
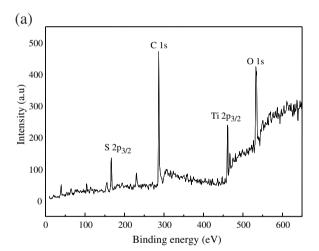


Fig. 4. UV-vis absorption spectra of P3MeT and P3MeT/TiO<sub>2</sub>.

signal from the thiophene rings in the ESCA spectra of P3MeT/TiO<sub>2</sub>/ITO at various depths. Letters A to G are the numbers of etchings of the sample by Ar ions, starting from the film surface. The lack of sulfur signal in curve G is probably due to the complete removal of P3MeT from the sample by Ar ions.



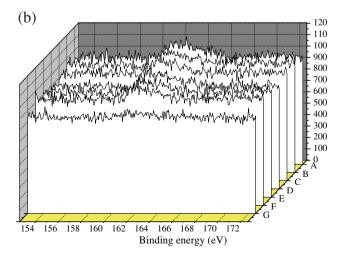


Fig. 5. ESCA spectra of (a) the surface of P3MeT/TiO $_2$  film, and (b) P3MeT/TiO $_2$ /ITO at various depths.

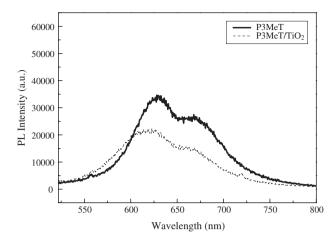


Fig. 6. PL spectra of P3MeT and P3MeT/TiO<sub>2</sub>.

Interestingly, the normalized intensity of the  $S(2p_{3/2})$  signal at 164 eV is almost the same for curves A to F, revealing that the composition ratio of  $TiO_2$  to P3MeT, is nearly fixed throughout the sample and the electropolymerization of 3-MeT in the  $TiO_2$  matrix probably began from the bottom ITO electrode, before the pores were filled. This also indicated that P3MeT chains were uniformly grown inside the porous  $TiO_2$  film. Furthermore, the thickness of the P3MeT/ $TiO_2$  film can be estimated from the etching depth before all of the P3MeT has been removed. It was determined to be approximately 60 nm, close to the value of 70 nm determined by the  $\alpha$ -stepper.

Charge transfer is known to normally occur with a very high efficiency if exitons are formed in conducting polymer within approximately  $10\sim20$  nm [13,14] of a titania interface. The PL measurements in Fig. 6 show that the exitons in the polymer are visibly quenched, suggesting that polymer chains were successfully synthesized inside the pores of  $\text{TiO}_2$  matrix, and electron transfer occurred from the polymer to  $\text{TiO}_2$ . The quenching efficiency of each film can be determined by calculating the integrated area beneath each curve [15]. The quenching efficiency of P3MeT/TiO<sub>2</sub> was found to be 31%.

The current-voltage characteristics of Al/P3HT/P3MeT/ TiO<sub>2</sub>/ITO device were measured under Xe-lamp illumination with an intensity of 88 mW/cm<sup>2</sup>. The short-circuit current, open-circuit voltage, fill factor and power conversion efficiency of the cell were determined to be 23.3 μA/cm<sup>2</sup>, 160 mV, 0.19 and a  $3.22 \times 10^{-3}$ %, respectively. Two possible reasons are responsible for the rather low quenching efficiency of the hybrid film and low power conversion efficiency of the cell. Firstly, the incompatibility of inorganic particles and organic polymers results in a poor contact between polymer and TiO<sub>2</sub> surface after electropolymerization process, reducing the efficiency of photo-induced electron transfer. Secondly, the pore size of the porous TiO<sub>2</sub> film used here is still small relative to the polymer chain-length, which causes a coiled and twisted chain conformation for the poly(3-hexylthiophene)s grown inside the pores, thus lowering the hole mobility and blueshifting the UV-vis absorption spectrum. However, this study provides a facile and successful route for growing conducting polymer from the porous TiO<sub>2</sub> film. The cell power conversion efficiency could be further improved by optimizing the in situ polymerization conditions, polymer structures, film morphologies, and contacts of metal/polymer/inorganic interfaces.

# 4. Conclusions

A method of developing an organic–inorganic hybrid as a photoactive material in ultrathin photovoltaic devices via in situ polymerization technique is described. UV–vis and photoluminescence spectroscopy measurements suggest that the polymer chains are coiled inside the nanoporous titanium films. ESCA spectra indicate the P3MeT chains were successfully electropolymerized and packed inside the porous  $TiO_2$  matrix. The photovoltaic cell fabricating from the prepared P3MeT/ $TiO_2$  film demonstrates an open-circuit voltage of 0.16 V, a short-circuit current density of 23.3  $\mu$ A/cm², a fill factor of 0.19, and a power conversion efficiency of  $3.22 \times 10^{-3}$ %.

### Acknowledgements

The authors would like to thank the National Science Council and the Ministry of Education of the Republic of China, Taiwan and US Air Force for financially supporting this research.

#### References

- [1] C.J. Brabec, N.S. Sariciftci, J.C. Hummelen, Adv. Funct. Mater. 3 (2000) 15.
- [2] W.U. Huynh, J.J. Dittmer, A.P. Alivisatos, Science 295 (2002) 2425.
- [3] S.E. Shaheen, C.J. Brabec, N.S. Sariciftci, Appl. Phys. Lett. 78 (2001) 841.
- [4] J.J.M. Halls, K. Pichler, R.H. Friend, S.C. Moratti, A.B. Holmes, Appl. Phys. Lett. 68 (1996) 3120.
- [5] K.E. Ziemelis, A.T. Hussain, D.D.C. Bradley, R.H. Friend, J. Rühe, G. Wegner, Phys. Rev. Lett. 66 (1991) 2231.
- [6] P. Dyreklev, O. Inganäs, J. Paloheimo, H. Stubb, J. Appl. Phys. 71 (1992) 2816.
- [7] T.J. Savenije, J.M. Warman, A. Goossens, Chem. Phys. Lett. 287 (1998)
- [8] F. Garnier, G. Horowitz, Synth. Met. 18 (1986) 693.
- [9] C.J. Wang, Y.R. Zhu, Z.Y. Chen, Mater. Sci. Eng. 77 (2000) 135.
- [10] G.R.A. Kumara, A. Konno, K. Shiratsuchi, J. Tsukahara, K. Tennakone, Chem. Mater. 14 (2002) 954.
- [11] B. O'Regan, D.T. Schwartz, S.M. Zakeeruddin, M. Grätzel, Adv. Mater. 12 (2000) 1263.
- [12] J. Krüger, R. Plass, M. Grätzel, H.J. Mathieu, Appl. Phys. Lett. 81 (2002) 367
- [13] N.C. Greenham, X. Peng, A.P. Alivisatos, Phys. Rev. B 54 (1996) 17628.
- [14] P.A. van Hal, M.P.T. Christiaans, M.M. Wienk, J.M. Kroon, R.A.J. Janssen, J. Phys. Chem. B 103 (1999) 4352.
- [15] K.M. Coakley, Y. Liu, M.D. McGehee, K.M. Frindell, G.D. Stucky, Adv. Funct. Mater. 13 (2003) 301.