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New non-fatigue ferroelectric thin films of barium bismuth tantalate

Chung-Hsin Lu ^{*}, Cheng-Yen Wen*Department of Chemical Engineering, National Taiwan University, Taipei, Taiwan*

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Abstract

New non-fatigue ferroelectric thin films of barium bismuth tantalate ($\text{BaBi}_2\text{Ta}_2\text{O}_9$) were synthesized in this work. These films were prepared on Pt/Ti/SiO₂/Si substrates by the metalorganic decomposition method. As-deposited films were amorphous, and became well-crystallized after annealing at 700°C. The annealed films exhibited fairly smooth surface and small grain size (around 10 nm). The measured dielectric constant and dissipation factor of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ films at 10 kHz were 97.7 and 0.0257, respectively. The polarization–electric field hysteresis loops revealed the ferroelectric characteristics of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ films. Furthermore, the fatigue test indicated that these films hardly degraded in the polarization after 10⁹ switching cycles. Because of its ferroelectric properties and excellent fatigue resistance, $\text{BaBi}_2\text{Ta}_2\text{O}_9$ has great potential in becoming a new candidate material for the applications of ferroelectric random access memories. © 1999 Elsevier Science B.V. All rights reserved.

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1. Introduction

Ferroelectric materials have attracted much attention for the applications to non-volatile random access memories [1]. Among these materials, $\text{Pb}(\text{Zr},\text{Ti})\text{O}_3$ has been intensively investigated for more than a decade. Although $\text{Pb}(\text{Zr},\text{Ti})\text{O}_3$ exhibits excellent ferroelectric properties, $\text{Pb}(\text{Zr},\text{Ti})\text{O}_3$ thin films fabricated on metal electrodes suffer serious degradation of remnant polarization after long-term switching cycles [2]. Various models concerning the mechanisms for the loss of polarization have been

proposed [3–5]. Oxygen vacancies are considered to be the main cause for such fatigue phenomenon. On the other hand, bismuth-layered perovskite $\text{SrBi}_2\text{Ta}_2\text{O}_9$ thin films have been reported to be a potential candidate for non-volatile random access memories, since these films exhibit good polarization hysteresis behavior and fatigue-free (up to 10¹² switching cycles) characteristics [6]. A lot of efforts have been devoted to integrate these materials for the application of VLSI [7–9]. These studies show that $\text{SrBi}_2\text{Ta}_2\text{O}_9$ exhibits low leakage current density, long data retention time, and stable imprint characteristics. Furthermore, $\text{SrBi}_2\text{Ta}_2\text{O}_9$ thin films have superior fatigue resistance compared to $\text{Pb}(\text{Zr},\text{Ti})\text{O}_3$ films.

^{*} Corresponding author. Fax: +886-2-362-3040

Although the preparation and properties of $\text{SrBi}_2\text{Ta}_2\text{O}_9$ films have been thoroughly studied, a similar compound $\text{BaBi}_2\text{Ta}_2\text{O}_9$ has not been investigated in detail. It was reported that $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films showed low leakage current and had good dielectric properties for the use in conventional circuits [10]. In the ceramic form, $\text{BaBi}_2\text{Ta}_2\text{O}_9$ possesses a lower Curie temperature, and a higher dielectric constant than does $\text{SrBi}_2\text{Ta}_2\text{O}_9$ at room temperature [11]. We have investigated the formation mechanism and thermal stability of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ ceramics, and proposed a new process different from the traditional solid-state reaction to prepare well-densified $\text{BaBi}_2\text{Ta}_2\text{O}_9$ ceramics [12]. On the other hand, the studies concerning the $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films are few. In this study, we used MOD (metallorganic decomposition) method to synthesize $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films on Pt/Ti/SiO₂/Si substrates. The phase evolution of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films during annealing processes was examined. The microstructures of the prepared films were observed by scanning electron microscopy (SEM) and atomic force microscopy (AFM). Furthermore, the dielectric and ferroelectric properties of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ films were investigated, including remnant polarization, coercive field, and fatigue endurance.

2. Experimental

Barium 2-ethylhexanoate, bismuth 2-ethylhexanoate, and tantalum ethoxide were used as the starting materials, and toluene as the solvent. The solutions were thoroughly mixed according to the stoichiometric ratio, then the prepared precursor was spin-coated onto Pt/Ti/SiO₂/Si substrates. The spin-coated films were heated on a hot plate at 150°C to remove residual toluene solvent and were subsequently pyrolyzed at 400°C. The pyrolyzed films were annealed at 500–800°C in the flowing oxygen for 2 h. The thickness of prepared $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films was approximately 0.4 μm. Phase identification was performed using X-ray diffraction (XRD). The surface morphology was observed by SEM and AFM using a tapping mode with amplitude modulation. The dielectric properties were measured on metal–ferroelectric–metal capacitors using an impedance analyzer. A standard ferroelec-

tricity analyzer was employed to detect the ferroelectric characteristics.

3. Results and discussion

Fig. 1 shows the XRD patterns of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films deposited on Pt/Ti/SiO₂/Si substrates at various annealing temperatures. At 500°C the film remains amorphous as shown in Fig. 1a. From 600°C (see Fig. 1b), $\text{BaBi}_2\text{Ta}_2\text{O}_9$ phase starts to form and the (105) diffraction peak appears at around $2\theta = 28^\circ$. However, the rather broad peak indicates that this film is not fully crystallized. The crystallization of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ films gradually progresses with increasing temperature. At 700°C the crystallization of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films is further enhanced, and the diffraction peaks of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ are clearly observed (see Fig. 1d), indicating that $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film is well-crystallized. The diffraction peaks in Fig. 1d are well in consistence with those of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ ceramics reported [12]. Therefore, it is confirmed that the $\text{BaBi}_2\text{Ta}_2\text{O}_9$ single phase was obtained. It is noted that no intermediate phase is found within the temperature range of 500–700°C.

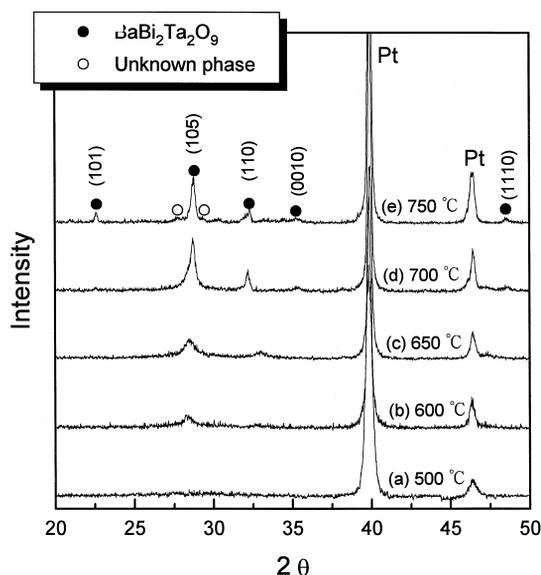


Fig. 1. XRD patterns of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films deposited on Pt/Ti/SiO₂/Si substrates after annealing at (a) 500, (b) 600, (c) 650, (d) 700 and (e) 750°C.

Apparently, the crystallized $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films are nucleated and grown directly from the amorphous phase. The lattice constants of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film are calculated from the d -spacings of peaks (101), (105), (110), and (0010). Assuming that the crystal system of the $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film is pseudo-tetragonal, the lattice constants are deter-

mined to be $a = 0.3942$ nm, and $c = 0.2542$ nm. On the other hand, a tiny amount of an unknown phase appears after 750°C annealing, as shown in Fig. 1e. This phase is probably formed from the interaction between films and substrates.

Fig. 2a shows the scanning electron microscopic image of the $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film annealed at

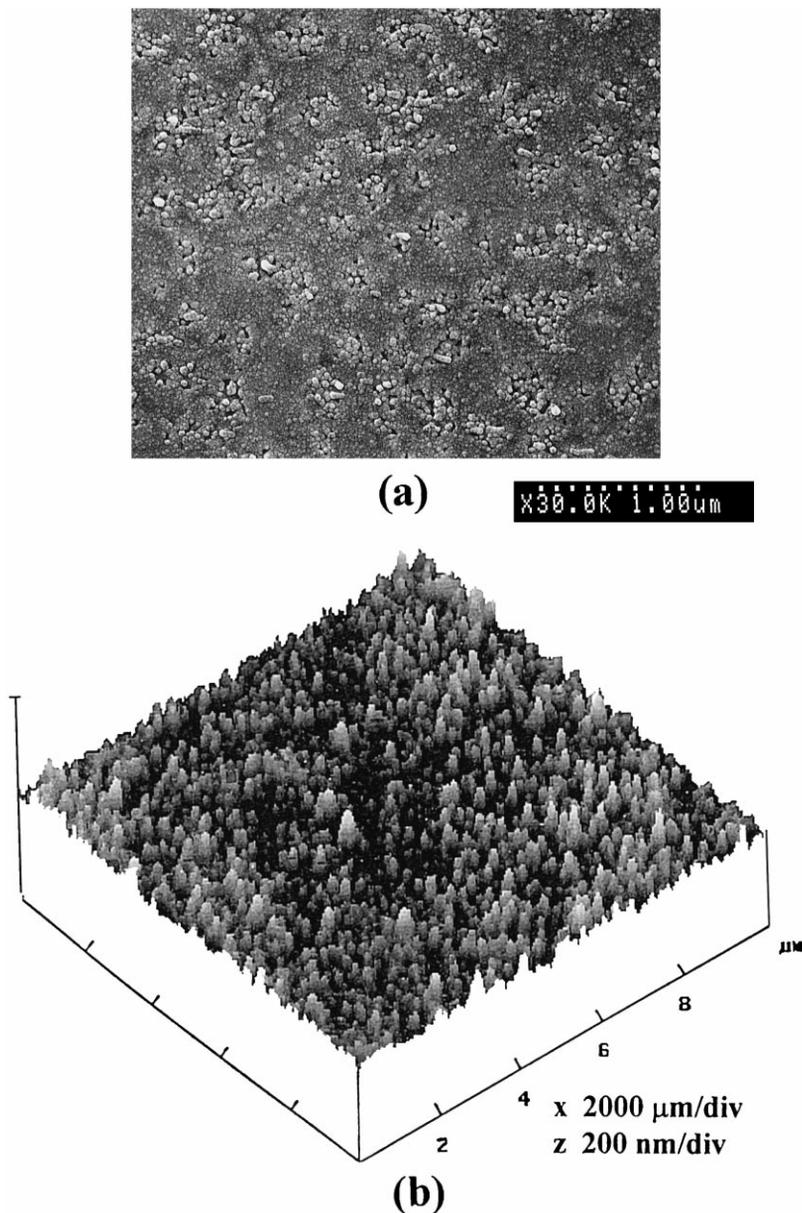


Fig. 2. (a) Scanning electron micrograph and (b) atomic force micrograph of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films annealed at 700°C .

700°C. It is found that the grains of this film have a spherical shape with small grain size, and the surface of this film seems dense and flat. Fig. 2b demonstrates the surface morphology of the same film observed by an atomic force microscope with scanning area of $10\ \mu\text{m} \times 10\ \mu\text{m}$. This image indicates that the surface of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film is fairly smooth, and no cracks are formed. The average surface roughness of this film is calculated to be less than 10 nm.

The dielectric properties of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films were measured. Fig. 3 shows the dielectric constant and the dissipation factor as a function of frequency for the 0.4- μm thick $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film with Pt/ $\text{BaBi}_2\text{Ta}_2\text{O}_9$ /Pt structure. The measuring frequency ranges from 100 Hz to 4 MHz. The dielectric constant and dissipation factor at 10 kHz are 97.7 and 0.0257, respectively. Within the low frequency range, the dissipation factor is close to zero, which is undetectable. With an increase in frequency, the dissipation factor slightly rises, indicating that the conductivity of the films is increased with the measuring frequency.

The polarization–electric field curves of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films annealed at 700 and 750°C are shown in Fig. 4. Hysteresis measurements were conducted under the operating voltage of 5 V, and the corresponding electrical field is 125 kV/cm. For

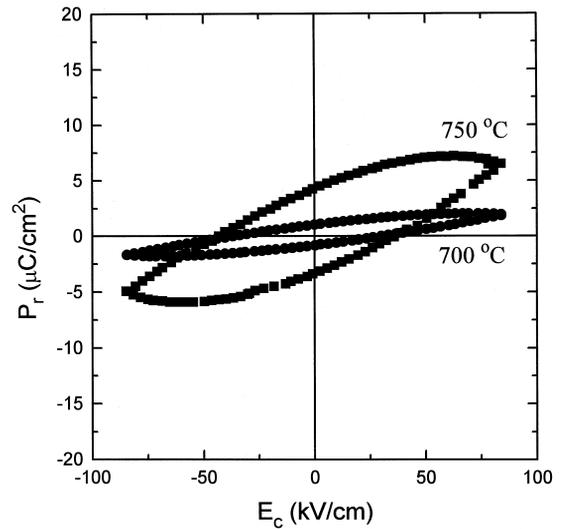


Fig. 4. Polarization electric field hysteresis curves of annealed $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films.

$\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin film annealed at 700°C, the remnant polarization ($2P_r$) and the coercive field ($2E_c$) are around $2\ \mu\text{C}/\text{cm}^2$ and 60 kV/cm, respectively. However, for the film annealed at 750°C, the remnant polarization ($2P_r$) and the coercive field ($2E_c$) are raised to $8\ \mu\text{C}/\text{cm}^2$ and 75 kV/cm. From the P – E curves, it is confirmed in this study that

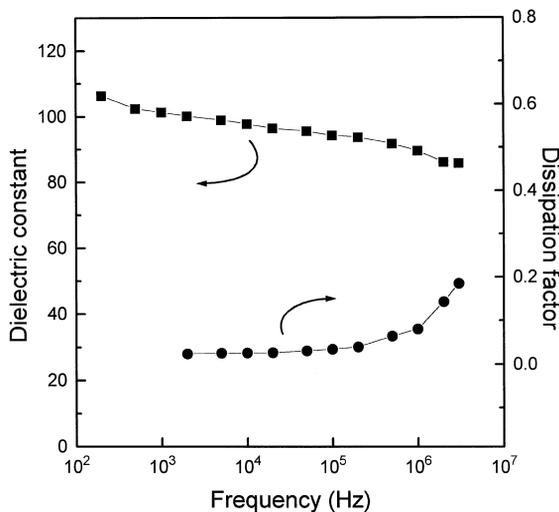


Fig. 3. Plots of dielectric constant and dissipation factor of $\text{BaBi}_2\text{Ta}_2\text{O}_9$ thin films as a function of measuring frequency.

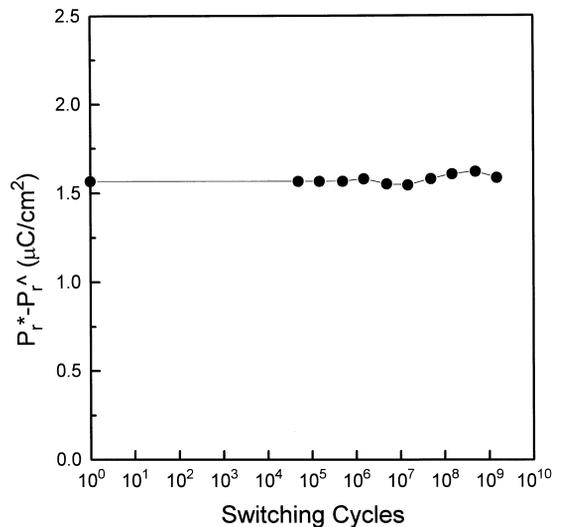


Fig. 5. Polarization fatigue endurance test for Pt/ $\text{BaBi}_2\text{Ta}_2\text{O}_9$ /Pt capacitors under 8.6- μs wide bipolar pulses of 5 V amplitude.

BaBi₂Ta₂O₉ thin films exhibit ferroelectric properties. Since the Curie temperature of BaBi₂Ta₂O₉ is 110°C [11], it is reasonable to predict that BaBi₂Ta₂O₉ exhibits ferroelectric properties at room temperature.

The fatigue endurance of BaBi₂Ta₂O₉ thin film capacitor as a function of switching cycles was examined by applying 8.6-μs wide bipolar pulses of 5 V amplitude, and the results are shown in Fig. 5. P_r^* and P_r^\wedge represent the values of the switched and non-switched remnant polarization, respectively. The value of $P_r^* - P_r^\wedge$ is approximately corresponding to the value of $2P_r$. Because the 750°C-annealed films contain a tiny amount of an unknown phase, only the pure BaBi₂Ta₂O₉ films annealed at 700°C were examined. It is found that the value of $P_r^* - P_r^\wedge$ does not decay after 10⁹ switching cycles, indicating that BaBi₂Ta₂O₉ thin films are near fatigue-free. As mentioned previously, Pb(Zr,Ti)O₃-based ferroelectric thin films suffer a serious fatigue problem when fabricated on metal electrodes. However, this study shows that BaBi₂Ta₂O₉ thin films deposited on metal electrodes have good polarization endurance as SrBi₂Ta₂O₉ thin films do. Because of its high fatigue-resistance, BaBi₂Ta₂O₉ is found in this study to be a potential material for the applications of ferroelectric random access memories.

4. Conclusion

(i) BaBi₂Ta₂O₉ thin films were successfully synthesized on Pt/Ti/SiO₂/Si substrates with the metalorganic decomposition method. Well-crystallized BaBi₂Ta₂O₉ thin films were obtained after 700°C-annealing.

(ii) The grains in the 700°C-annealed BaBi₂Ta₂O₉ films had a small grain size (around 10 nm) with a spherical shape. In addition, the surface of the prepared films was rather dense and smooth with no cracks.

(iii) The dielectric constant and dissipation factor of BaBi₂Ta₂O₉ films at a frequency of 10 kHz were 97.7 and 0.0257, respectively. The polarization–electric field hysteresis loops revealed that BaBi₂Ta₂O₉ films exhibited ferroelectric properties, and the loss in the remnant polarization of the films was near zero up to 10⁹ switching cycles. According to the above results, BaBi₂Ta₂O₉ has been proved to have great potential for the applications of ferroelectric random access memories.

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