

Diffusion and crystallization mechanisms of Ge/Au bilayer media for write-once optical disk

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Ge/Au bilayer thin films were fabricated by magnetron sputtering method, the temperature dependence of resistance from room temperature to 500 °C and concentration depth profiles are measured. From the temperature dependence of resistance measurement, we found two phase change phenomena which occurred at 175 and 360 °C. The element concentration depth profiles of the as-deposited and recorded region indicate that the Au-Ge alloy is initially formed at the Ge/Au interface. The dominant diffusion element is Au atom and the diffusion path is from Au layer to Ge layer. The optimum simulated bit error rate value is about 1.4×10^{-6} at 9.0 mW under two time high definition digital versatile disk (HD DVD) recording speed. The dynamic tests show that this Ge/Au bilayer films can be applied to one to two times HD DVD-R. © 2008 American Institute of Physics. [DOI: 10.1063/1.2831690]

Recently, the optical disk was improved due to the needs of large information transformation and high definition media resolution. Blu-ray disc (BD) and high definition digital versatile disk (HD DVD) had been proposed as the promising candidates. Various inorganic recording media were reported for them. The jitter value of Bi-Ge nitride¹ was found only 5.7% under 36–72 Mbps writing speed. Te-O-Pd (Ref. 2) was proposed as a four layer recording media for 100 Gbyte BD system. The bilayer structure, such as amorphous silicon (*a*-Si)/metal^{3–6} have also been studied. Although the relationship between microstructure and annealing temperature of *a*-Ge/Au had been discussed previously,⁵ the diffusion mechanism of Ge/Au bilayer during recording is not identified.

In this paper, the diffusion and crystallization mechanisms of *a*-Ge/Au bilayer media for write-once HD DVD disks are investigated. Two steps of phase change are found by measuring the variation of electrical resistance with temperature. The element concentration depth profiles of the as-deposited and recorded region, which were measured by Auger electron spectrometer (AES), indicate that the diffusion path was from Au layer to Ge layer.

The resistance of Ge/Au bilayer with nature oxidized (110) silicon wafer substrate was measured by the four-probe resistance tester in vacuum. Figure 1(a) shows the layer structure of the sample. The *a*-Ge(10 nm)/Au(10 nm) bilayer for AES investigation was prepared by magnetron sputtering on polycarbonated (PC) substrate, as shown in Fig. 1(b). The sample with PC substrate was then laser initialized, and the variation of its reflectivity with wavelength was mea-

sured by spectrometer (ETA-RT, Optik's high sensitive spectrometer). The initialized sample, which has the largest optical contrast at $\lambda=405$ nm (comparing with the as-deposited sample), is selected for the AES measurement. The element concentration depth profile was measured by AES with sputtering speed of 0.2 nm/s. The recording characteristics were evaluated by dynamic tester (ODU1000, PULSTEC), which the testing conditions are shown in Table I.

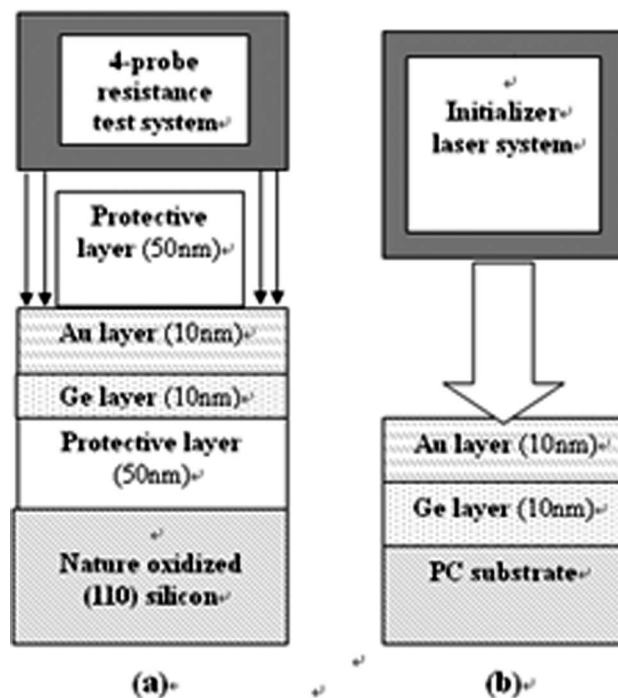


FIG. 1. The layer structure of samples for (a) resistance measurement and (b) AES depth profile measurement.

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TABLE I. Dynamic test conditions.

User capacity	Thickness of substrate	Wavelength	Numerical aperture (NA)	Modulation code	Track pitch	Channel clock frequency	Linear velocity	User bit rate
15 GB	0.6 mm	405 nm	0.65	ETM, RLL (1,10)	0.4 μm	64.8 Mhz	13.2 m/s	73.1 Mbps

Figure 2 shows the variation of resistance with temperature. The Ge/Au bilayer film has two phase transformation temperatures which are 175 and 360 °C. The first phase change temperature corresponds to the interdiffusion of the bilayers.⁵ Since the probes touch the surface of Au layer, as shown in Fig. 1(a), the measured resistance represents Au layer's behavior. The resistance increases with decreasing Au layer's thickness (Ohm's law, $R=\rho L/A$, where ρ =resistivity, L =sample length, and A =cross sectional area). The expansion of Ge/Au interface layer and the new forming discontinuous phase by interdiffusion of Au and Ge atoms resulted in the rising of resistivity at 175 °C. So, the crystallization temperature of Ge is decreased from 473 (Ref. 7) to 175 °C. This mechanism is called the metal induced crystallization (MIC) mechanism.^{3-6,8} From the observation of high resolution transmission electron microscope,⁵ rearranging atoms of the breaking of Au-Ge bonds and precipitation of Ge were found at 340 °C. However, from our resistance measurement, it shows no further phase change higher than 175 °C, until the temperature approaches to the eutectic temperature of Ge/Au (363 °C).⁹ This owes to the low solubility of Au in crystalline Ge (13.6 ppm),¹⁰ the metal induced crystallites of Ge hinder the further diffusion of Au into Ge at temperature between 175 and 360 °C, until the temperature approaches to the eutectic temperature of Ge/Au (363 °C), the Ge/Au films show another phase transformation.

In order to identify the atom's migration direction between Au and Ge layers during diffusion, the element depth profiles of as-deposited and recorded samples were measured by AES. Figure 3 shows the depth profile of Ge/Au bilayer. The total sputtering time is 100 s, which corresponds to about 20 nm depth from the surface. The film was composed

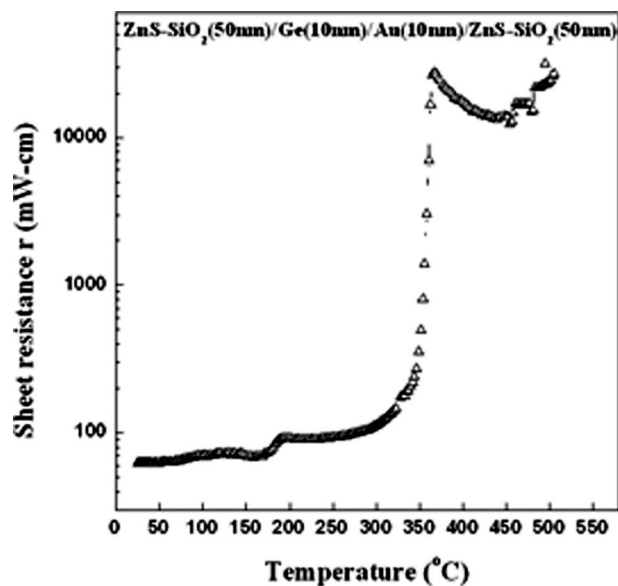


FIG. 2. Relationship between the electric resistance and temperature of Ge/Au bilayer.

by three layers, which are the Au layer, the Ge/Au interface layer, and the Ge layer. Significantly, the Au-Ge interlayer was formed initially. In 1989, Hou *et al.*¹¹ had discussed the heterogeneous nucleation and the random found limited mass transport at Au/Ge interlayer. In our examination, the diffusion path has been further revealed

Figures 4 and 5 are the Ge and Au concentration depth profiles of as-deposited and laser initialized in the Ge/Au bilayers, respectively. During the first 30 s of the sputtering (about 0–6 nm depth), the two curves in Fig. 4 remain extremely low. The Ge atoms did not diffuse to such depth after laser initialized. However, between 30 and 40 s (about 6–8 nm depth), the Ge concentration in laser recorded sample is slightly larger than that of the as-deposited sample. After that, although these two curves increase with the depth, the Ge concentration of laser initialized sample is lower than that of as-deposited sample. On the other hand, the Au atoms diffused into Ge layer as the depth deeper than 16 nm (sputtering time is longer than 80 s), as shown in Fig. 5. At the sputtering time of 30–40 s (about 6–8 nm depth), the Au concentration of laser recorded sample is slightly smaller than that of the as-deposited sample. These two curves decrease further as the sputtering time is increased again. However, the Au concentration of laser initialized sample is higher than that of as-deposited sample at sputtering time of 40–80 s (depth of 8–16 nm). Those results indicate that the Ge atoms did not obviously diffuse to the Au layer, the dominant diffusion element is Au and the diffusion path is from Au layer to Ge layer during recording. Au-Ge bonds were formed by Au interdiffusion. The broken metastable Au-Ge bonds were established during recording facilitated the local rearrangement.¹² Since the solubility of Au in crystallized Ge is low, the Au atoms will be squeezed out as the Ge crystallizes growing. The dependence between the metal thickness and the *a*-Ge layer was discussed.¹³ The MIC mechanism only affected the metal/*a*-Ge interface which was located

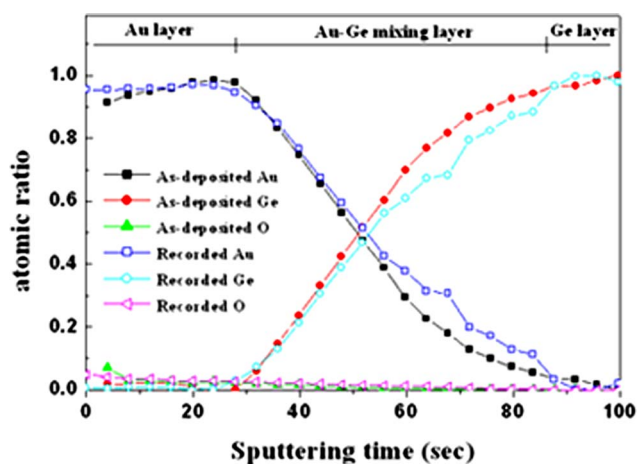


FIG. 3. (Color online) The concentration depth profile of as-deposited and laser initialized Ge/Au bilayer samples.

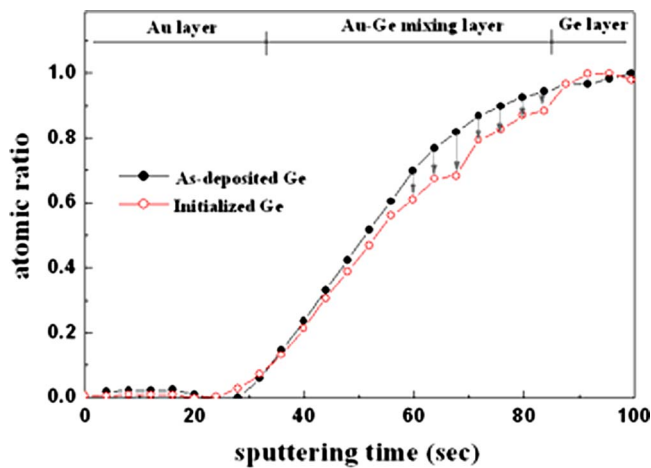


FIG. 4. (Color online) The concentration depth profile of as-deposited and laser initialized Ge atoms in the Ge/Au bilayer samples.

close to the bilayers. By the AES result, the diffusion limitation is revealed. Au atoms diffused into 8 nm depth of Ge layer, which resulted in the optical contrast between as-deposited and laser pulsed state.

Figure 6 shows the simulated bit error rate (Sber) value⁵ and the modulation of the Ge/Au bilayer compared with the laser powers. The Sber value is decreased from 1.2×10^{-5} to 1.4×10^{-6} as the writing power increased from 8.3 to 9.0 mW, and then further increased to 2.0×10^{-5} as the writing power increased to 9.6 mW. Meanwhile, the modulation is about 0.77–0.78. Sber value was suggested to be lower than 1×10^{-5} . The optimum Sber value in our dynamic test is 1.4×10^{-6} which occurred at about 9.0 mW under two times HD DVD recording speed that means the Ge/Au bilayer recording thin film is suitable for two times HD DVD write-once disk.

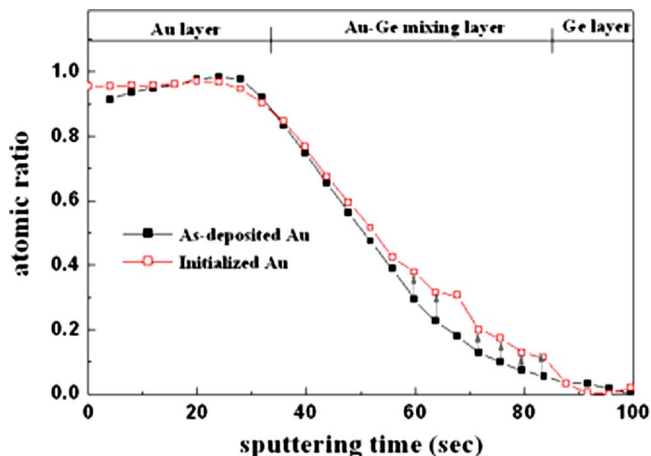


FIG. 5. (Color online) The concentration depth profile of as-deposited and laser initialized Au atoms in the Ge/Au bilayer samples.

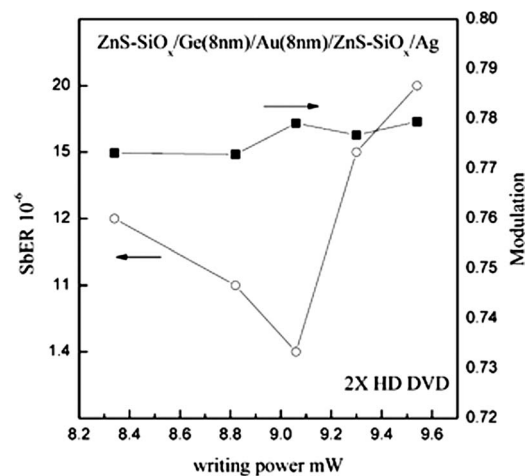


FIG. 6. Sber value and modulation of the Ge/Au bilayer medium as a function of writing power. The track pitch is $0.4 \mu\text{m}$.

In summary, recording properties of *a*-Ge/Au write-once optical media had been studied. From the four-probe resistance measurement, it is found that the crystallization temperature of Ge will be reduced from 473 to 175 °C by the interdiffusion of Au atoms. The AES inspection proves that the dominant diffusion element is Au and the diffusion path is from Au layer to Ge layer during recording. Dynamic tests show that the optimum Sber is 1.4×10^{-6} which occurred at about 9.0 mW under two times HD-DVD recording speed.

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